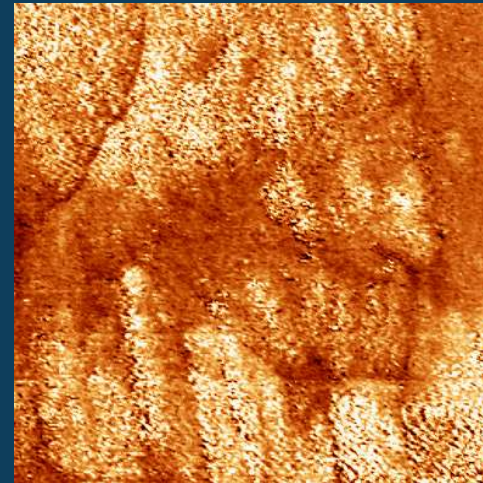
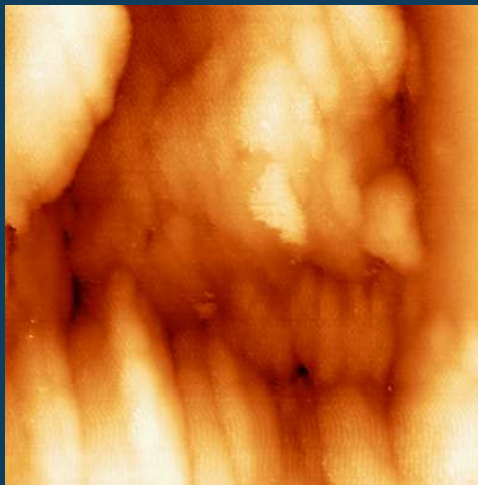


# Investigating the Insulator to Metal Transition in CMR $\text{La}_{0.67}\text{Ca}_{0.33}\text{MnO}_3$ via Scanning Tunneling Spectroscopy.



# Coupled Metal-Insulator/Ferromagnetic-Paramagnetic Transition in CMR Manganites

=> mixed valence via doping ( $\text{Mn}^{3+}$  &  $\text{Mn}^{4+}$ )

## Zero field transport:

### Above $T_c$

=> electron scattering dominated by polarons (e-ph. coupling mediated by J-T distortion at  $\text{Mn}^{3+}$ )

=> paramagnetism

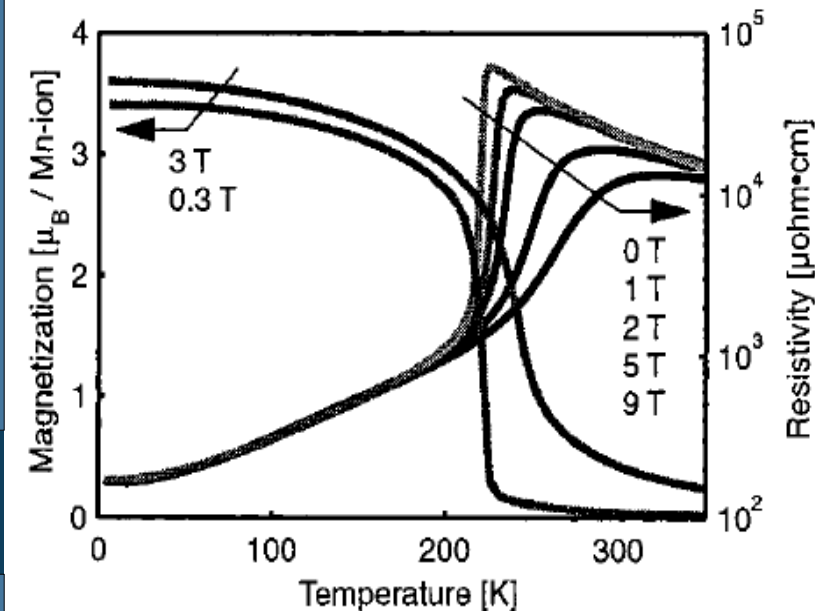
### Below $T_c$

=> metallicity

=> ferromagnetism  $\leftrightarrow$  double exchange

## Colossal Magneto Resistance (CMR)

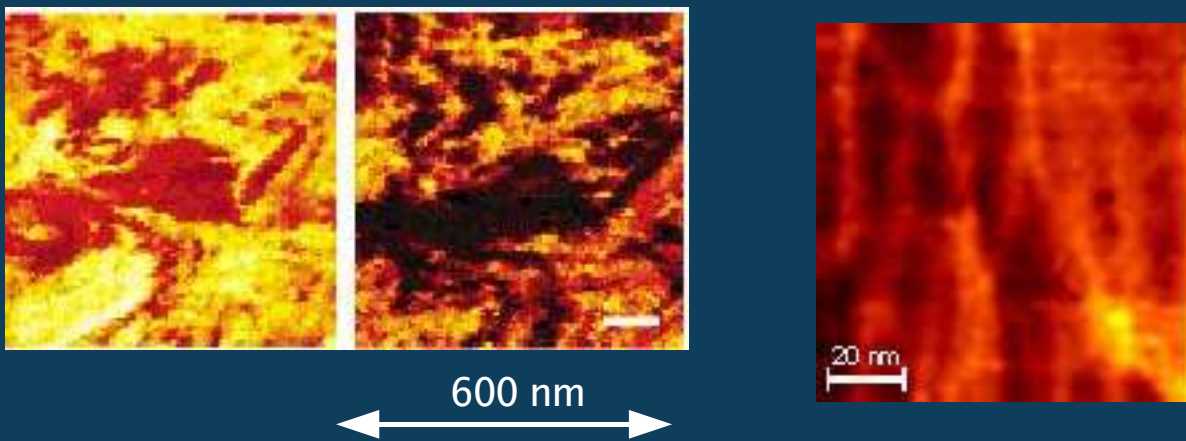
Applied magnetic field aligns spins, decreases PM (insulating) & increases FM (metallic). The effect changes resistance of several decades.



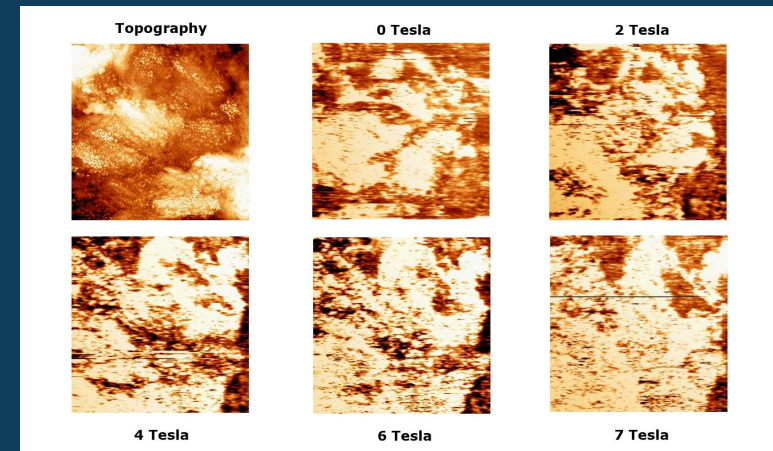
# Engineering the Electronic Properties Locally

## Spatial Electronic Phase Separation (PS)

- => Coexisting but spatially distinct insulating & metallic domains
- => Associated with *strained manganite films* grown on lattice mismatched substrates such as  $\text{LaAlO}_3$  (~2.3%),  $\text{SrTiO}_3$  (~-0.6%), and  $\text{MgO}$  (~8%).
- => *Strain free* films ( $\text{NdGaO}_3$  has a ~0.23% lattice mismatch with  $\text{La}_{0.7}\text{Ca}_{0.3}\text{MnO}_3$ ) are considered homogeneous (no PS).



M. Fäth et al. Science 285, 1540 (1999)



F.Galli et al. To be published.

=> **Influencing/inducing phase separations with artificial structures**

# Motivation

=> What happens to the local DOS while going through such a dramatic phase transition (homogeneous systems)? STM/STS probes **locally** the DOS.

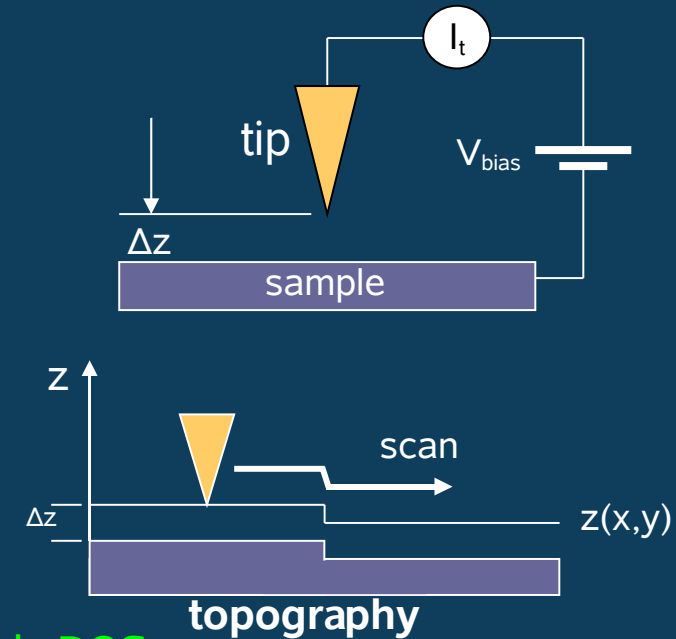
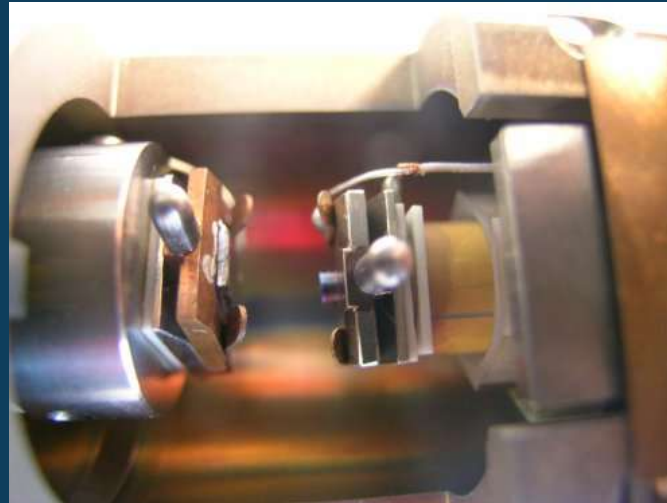
=> Finger print of the metallic phase in order to carefully study the homogeneities (or inhomogeneities) of the low temperature (metallic) phase.

=> DOS mapping at low bias  $V$  (0-500mV or less) where maximum change is seen in the MIT, versus high (1-2V) biases (less sensitive to changes in the MIT).

# Scanning Tunneling Microscopy and Spectroscopy

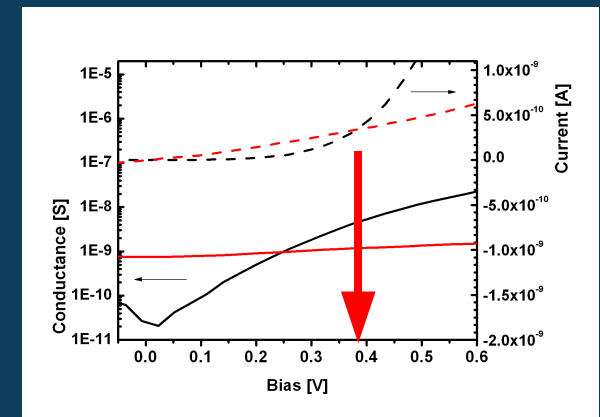
## Variable Temperature He-gas flow STM

- => 4K to 300K
- => 0 to +/- 8 T
- => home made STM
- => range:  
RT: 3.6 $\mu$ m or 1.5mm  
4K: 0.7 $\mu$ m or 0.3mm



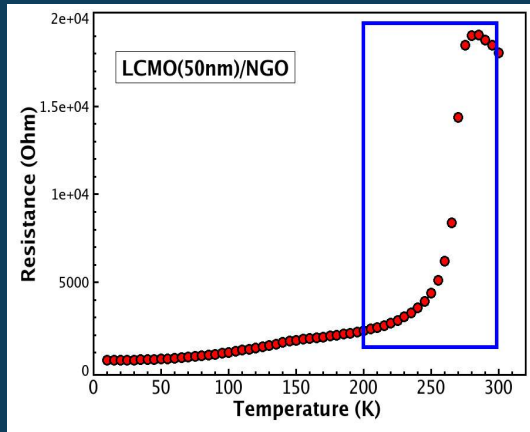
## Scanning Tunneling Spectroscopy (STS) → measure sample DOS

- scan topography
- at each pixel take I-V
  - stop scanning → x,y fixed
  - feedback off →  $\Delta z$  fixed
  - sweep bias  $\pm V$
  - measure  $I_t$
  - differentiate I-V curve for  $dI/dV$
  - Alternatively, measure directly  $dI/dV$  with lock-in technique.
  - No scanning, point spectroscopy.

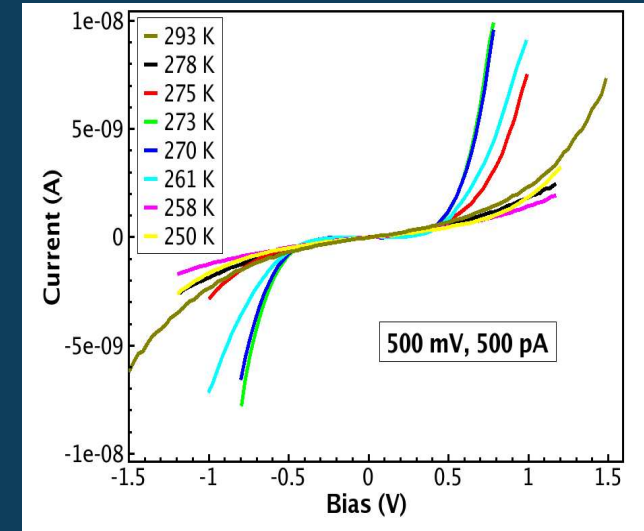


# La<sub>0.7</sub>Ca<sub>0.3</sub>MnO<sub>3</sub> (50nm Film) on NdGaO<sub>3</sub> Substrate Matching => Strain Free

## R-T Transport Measurement



=> Sharp transition at  $T_p=280$  K  
indicative of unstrained and  
electronically homogeneous film  
as expected for LCMO on  
lattice matched NGO substrate



## Temperature Dependent Point Spectroscopy

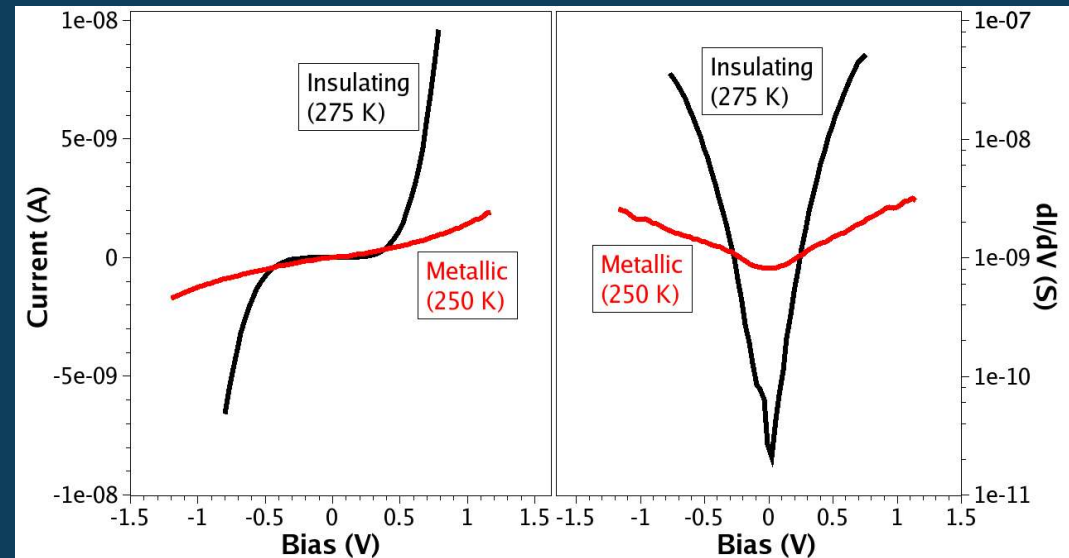
=>Low Bias to study changes of DOS around  $E_F$

=>I-V curves average of 5-20 I-V sweeps

=>I-V data shows metallic behavior well below  $T_p$   
changing to insulating character around  $T_p$

=>Metallic-like I-V curve zero-bias slope  
indicates significant conductance

=>Insulating-like I-V curves are flat around zero  
bias, indicative of gap-like character



# Tunneling Barrier Contribution (LCMO(50nm)/NGO)

=>Tip-Sample tunneling current across a tunneling barrier: barrier,  $t$ , sample DOS,  $N_s$ , and tip DOS,  $N_t$

$$I(s, V, W, T) = c \int_{-\infty}^{\infty} N_s(E + \frac{eV}{2}) N_t(E - \frac{eV}{2}) t(s, E, W) \times [f(E - \frac{eV}{2}) - f(E + \frac{eV}{2})] dE$$

=>Let's assume  $N_t$  and  $N_s$  as being both "flat" and try to deconvolute tunneling barrier part. Any deviation from a flat DOS will represent an energy dependence of the DOS itself. At low temperatures, we can approximate the Fermi-Dirac distribution as a step function, and simplify the above relation as:

$$I(s, V, W) \propto N_s N_t \int_{-\frac{eV}{2}}^{\frac{eV}{2}} t(s, E, W) dE$$

=>Model the tunneling barrier as one-dimensional and trapezoidal,

$$t(s, E, W) = e^{[-2ks\sqrt{2(W-E)}]}$$

=>Now we can relate conductance,  $dI/dV$ , to the tunneling barrier,

$$\frac{dI}{dV} \propto N_t N_s [t(s, \frac{eV}{2}, W) + t(s, -\frac{eV}{2}, W)]$$

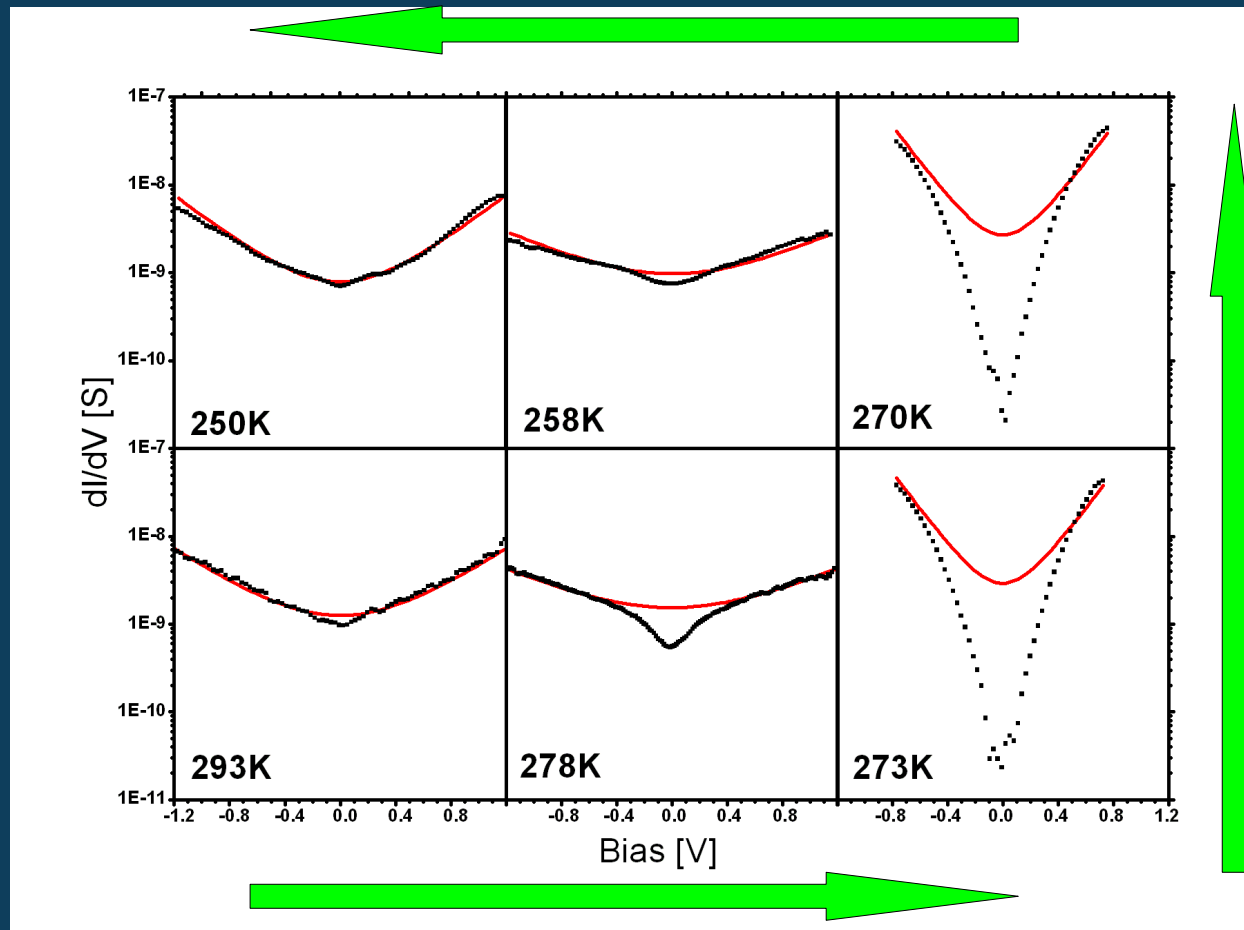
=>If we fit this function to our conductance data, we can estimate the tip-sample distance,  $s$  ( $\sim 10\text{\AA}$ ), and the work function,  $W$  ( $\sim 1-4\text{eV}$ ), and use these parameters to calculate the tunneling barrier as a function of temperature.

# dI/dV with Tunneling Barrier Fit (LCMO(50nm)/NGO)

=> dI/dV computed from I-V data (numerically differentiate).

=> Tunneling barrier fit (red) using higher bias data.

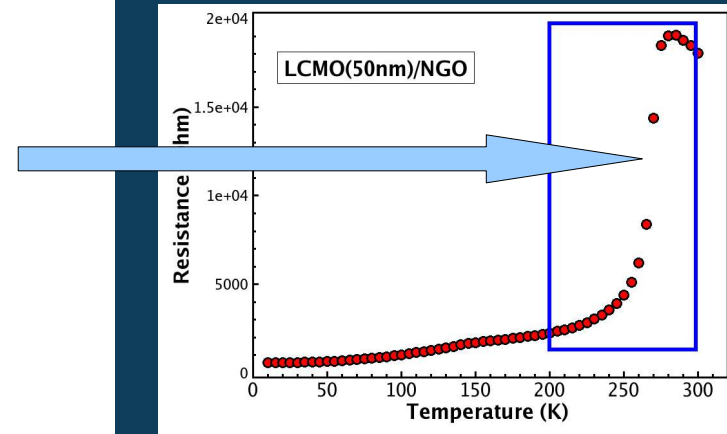
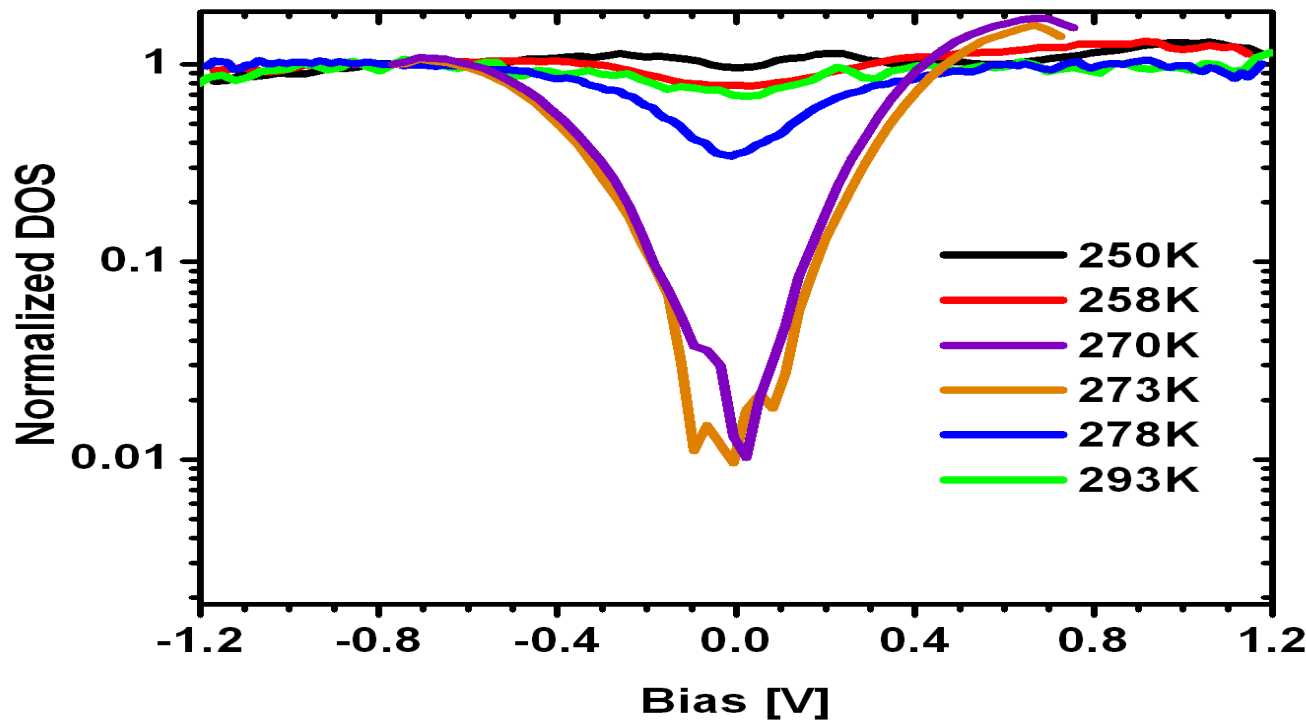
=> Difference clearly shows depletion of DOS in transition region is not associated with tunneling barrier effect





# Normalized dI/dV: Temperature Dependent DOS (LCMO(50nm)/NGO)

$$N_s \propto \frac{\frac{dI}{dV}}{N_t [t(s, \frac{eV}{2}, W) + t(s, -\frac{eV}{2}, W)]}$$



- => DOS indicates depletion, maximum below  $T_p$ , filling up again in the tail of the transition.
- => Temperature dependent *Coulomb gap? Pseudo gap* -precursor of metallic phase- ?
- => Only reported before by Mitra et al., PRB 71, 094426 (2005)

# Conclusions

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=> Observed strong temperature dependence of DOS via Tunneling Spectroscopy at low bias voltage.

=> Observe **depletion of electronic states** around  $T_p$ . Before the onset of the metallic phase the depletion gap fills up.

=> Depletion provides mechanism for significant conductance drop. Filling up of the gap is precursor of the metallic phase.