

Spin flip lasers

Tim Verhagen, Stefano Voltan & Hiske Overweg

Groupmeeting, May 4th 2011

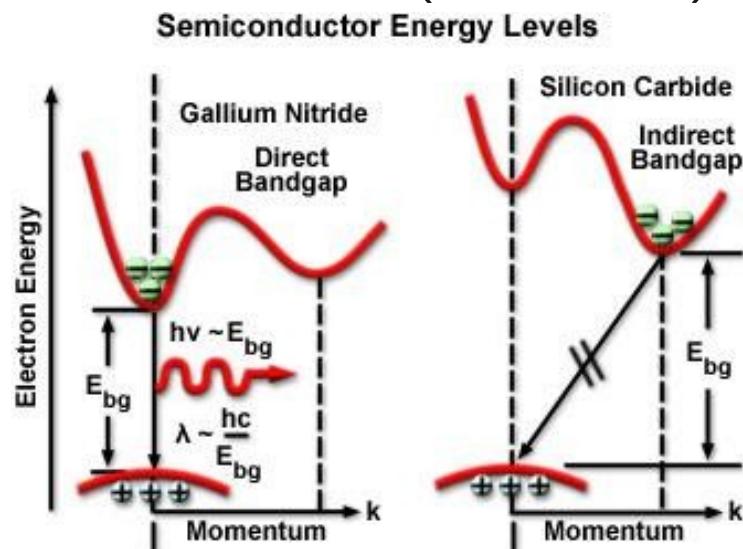
Background

Point contact spectroscopy

Hard magnets

Spin flip laser

Semiconductor (laser diode)



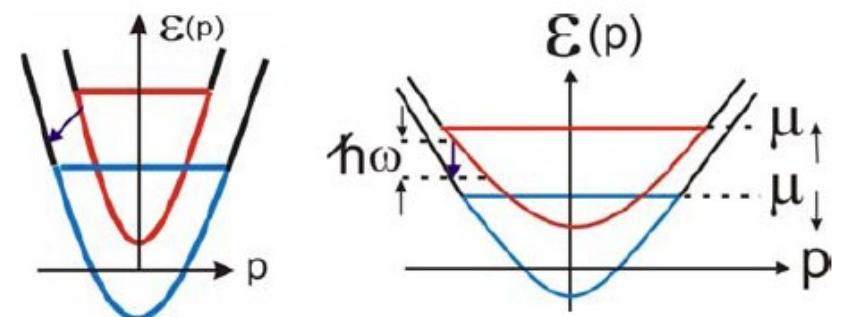
$$\Delta k = 0$$

photon

$$\Delta k \neq 0$$

phonon

Magnetic (spin flip laser)



$$\Delta k \neq 0$$

magnon

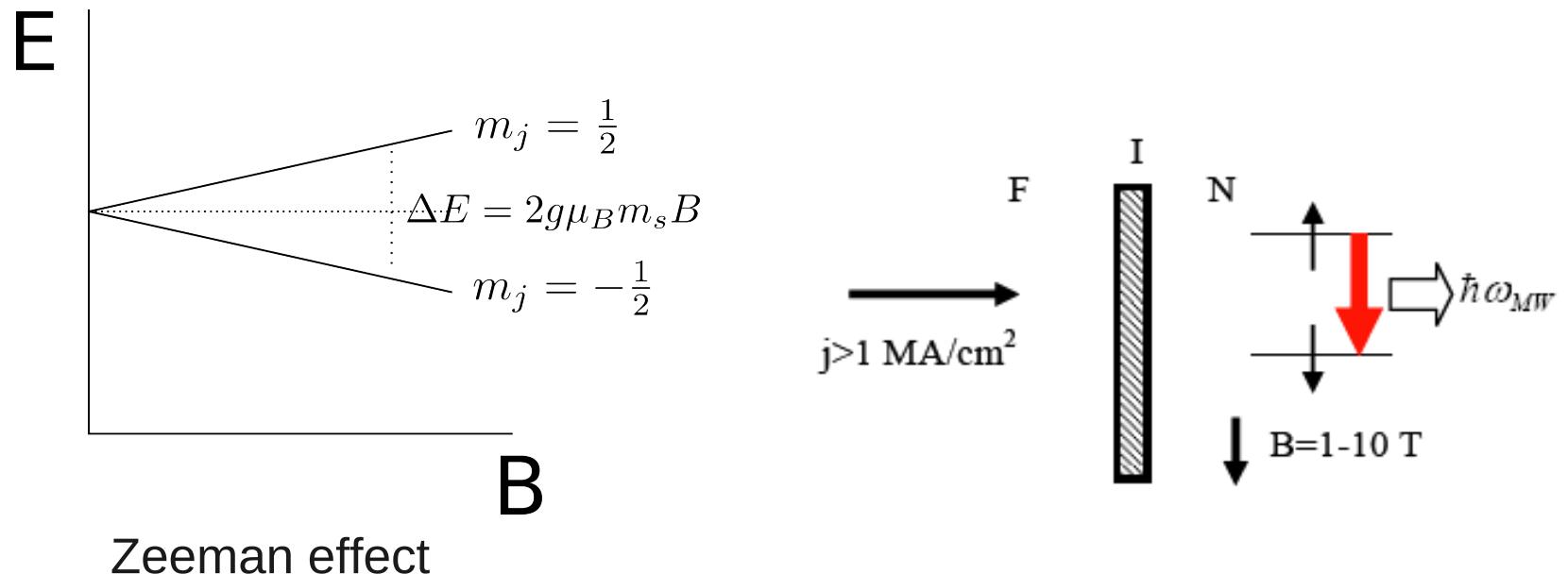
High frequency
oscillators

$$\Delta k = 0$$

photon

High frequency
lasers

Zeeman split transition



$$\begin{aligned}h\nu &= 2g\mu_B m_s B \\ \nu &= 0.014gB \left[\frac{\text{THz}}{T} \right]\end{aligned}$$

Devices, Majority F

If coercive field of F is (much) bigger than applied field

$\text{SmCo}_5, \text{AlNiCo}, \text{Nd}_2\text{Fe}_{14}\text{B}$

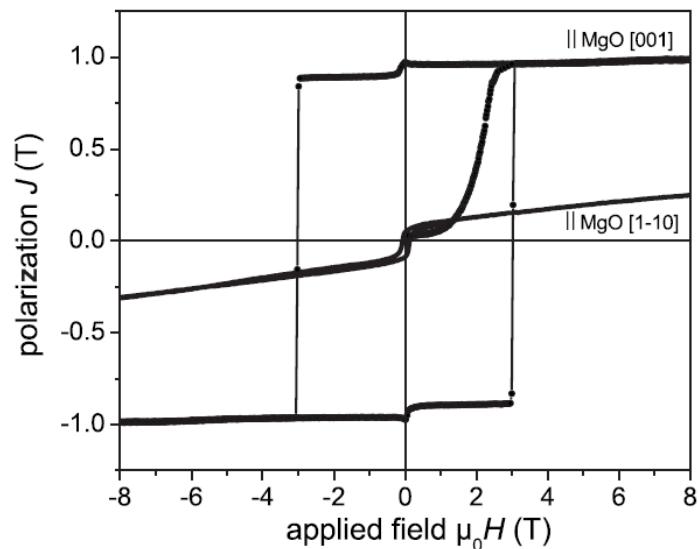


FIG. 2. Magnetic hysteresis of a SmCo_5 film measured along the easy magnetization axis (||MgO[001]) and along the in-plane hard axis (||MgO[1–10]).

Why magnetic lasers?

Electronics

$$P \propto \frac{1}{\nu^4}$$

Optics

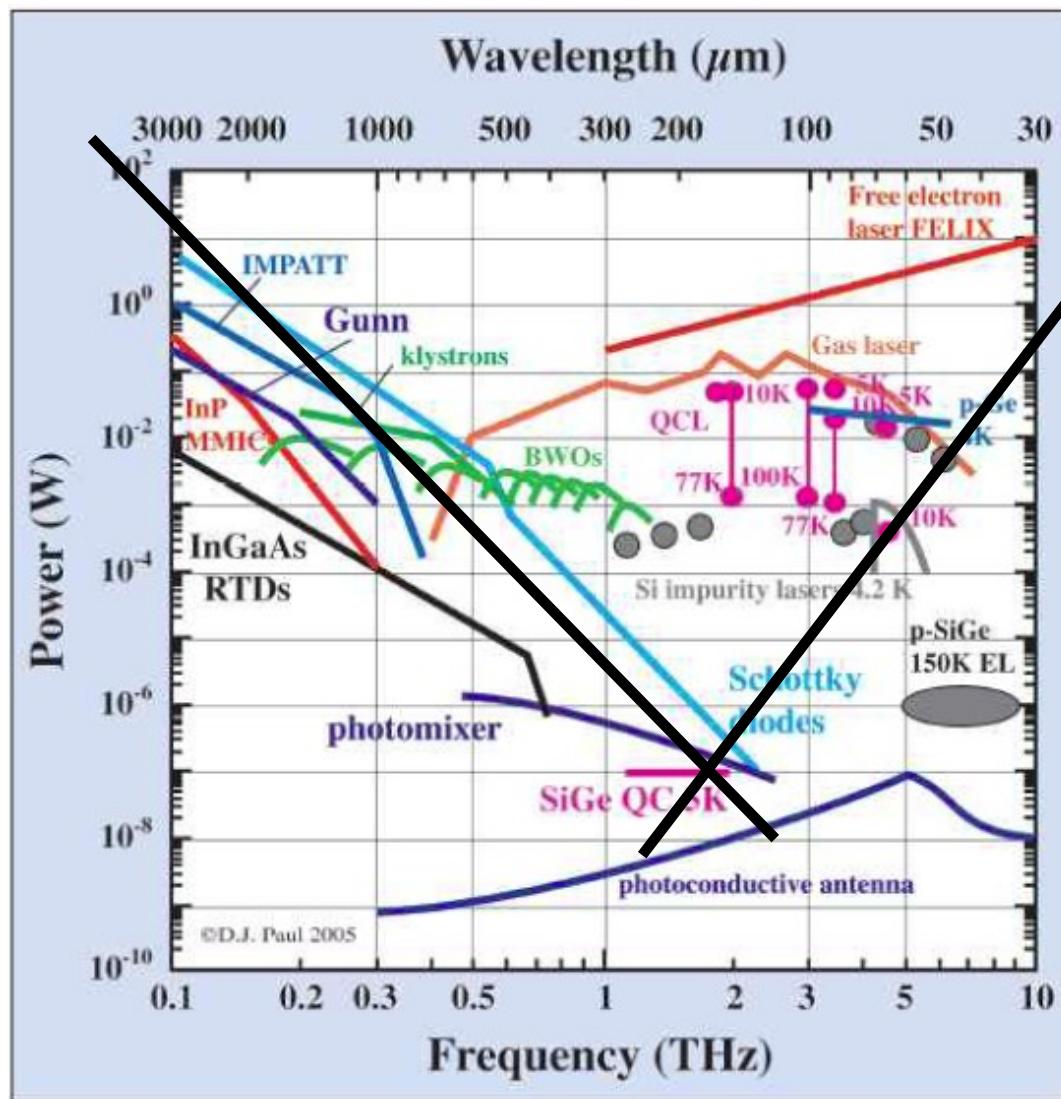
$$k_b T < h\nu$$

$$1 \text{ THz} = 4 \text{ meV}$$

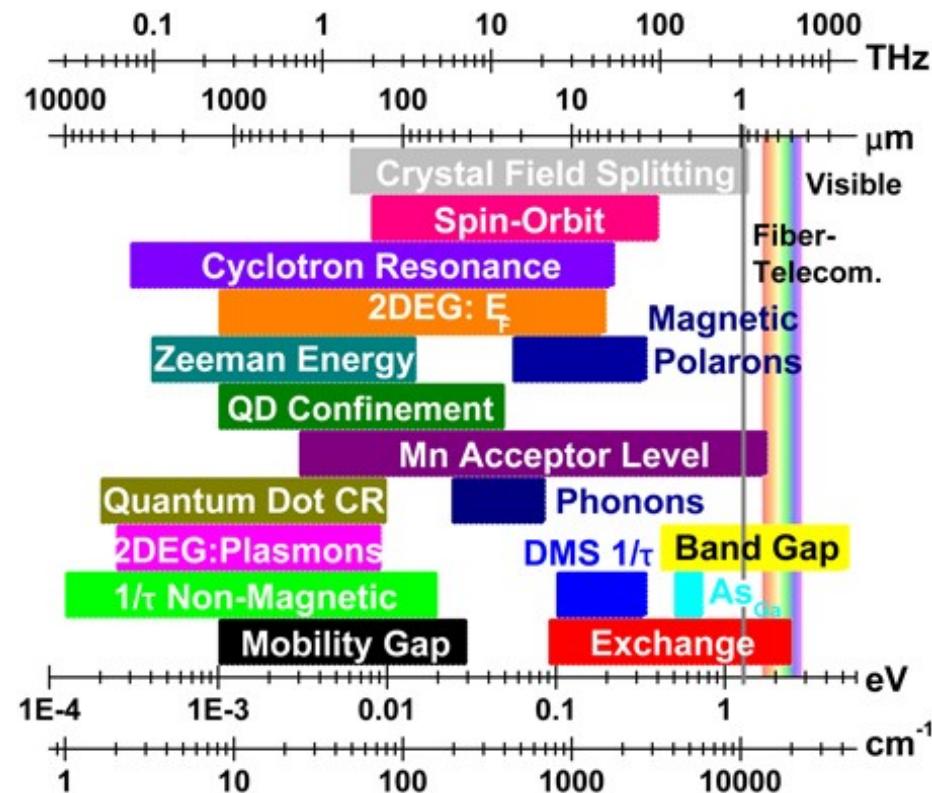
$$1 \text{ THz} = 48 \text{ K}$$

$$1 \text{ THz} = 33 \text{ cm}^{-1}$$

$$1 \text{ THz} = 0.3 \text{ mm}$$



What physics at THz frequencies?

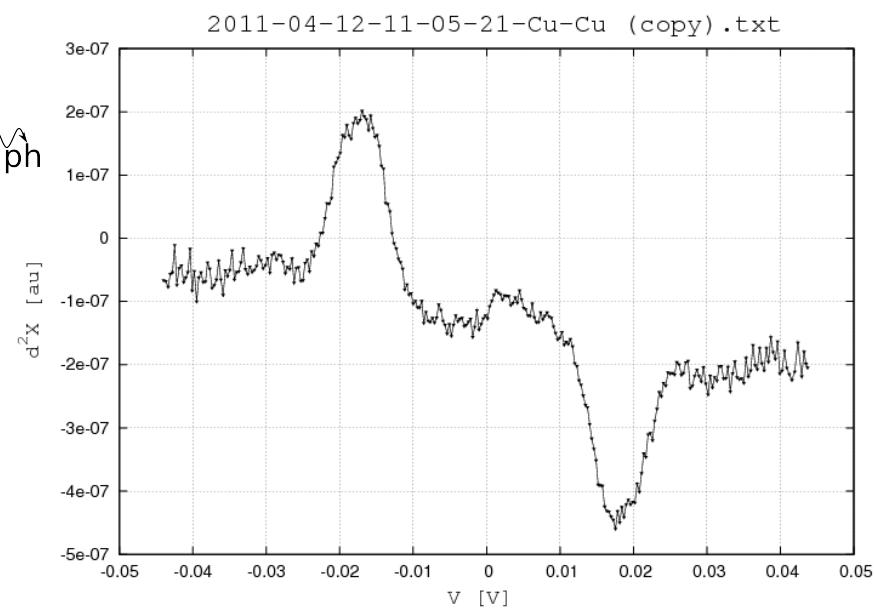
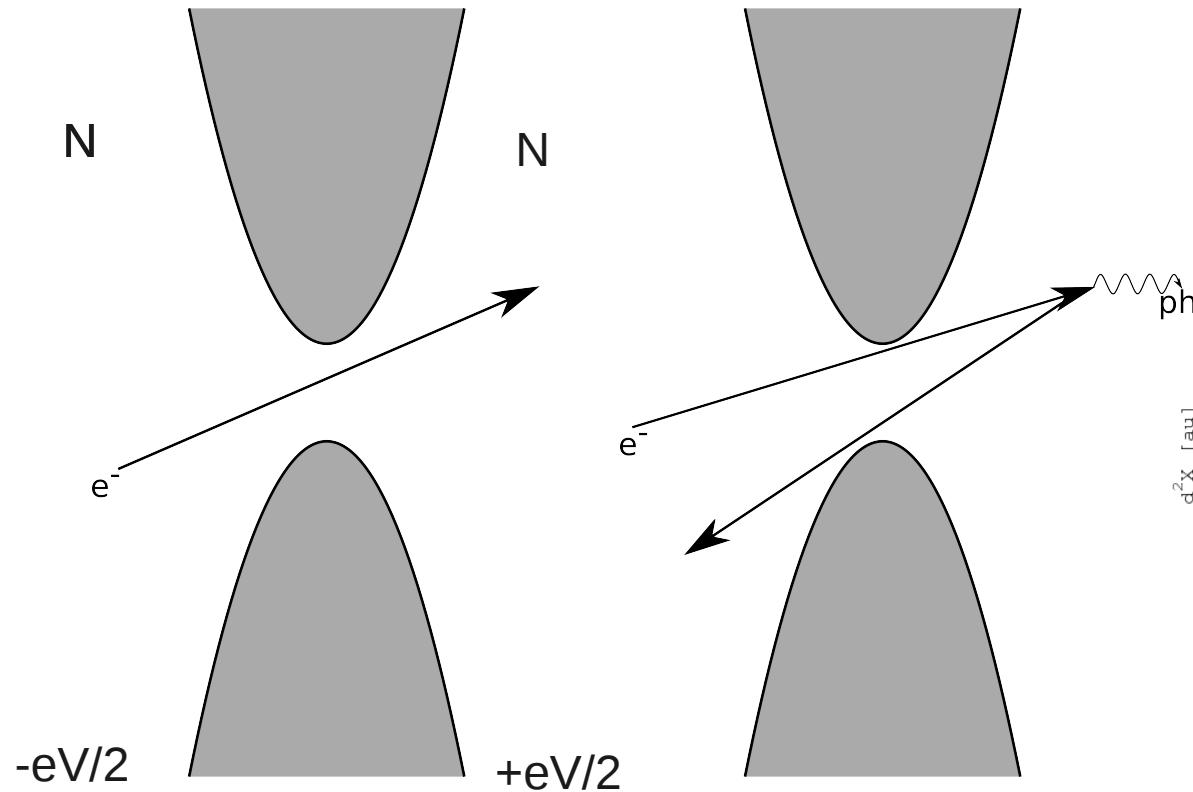


Background

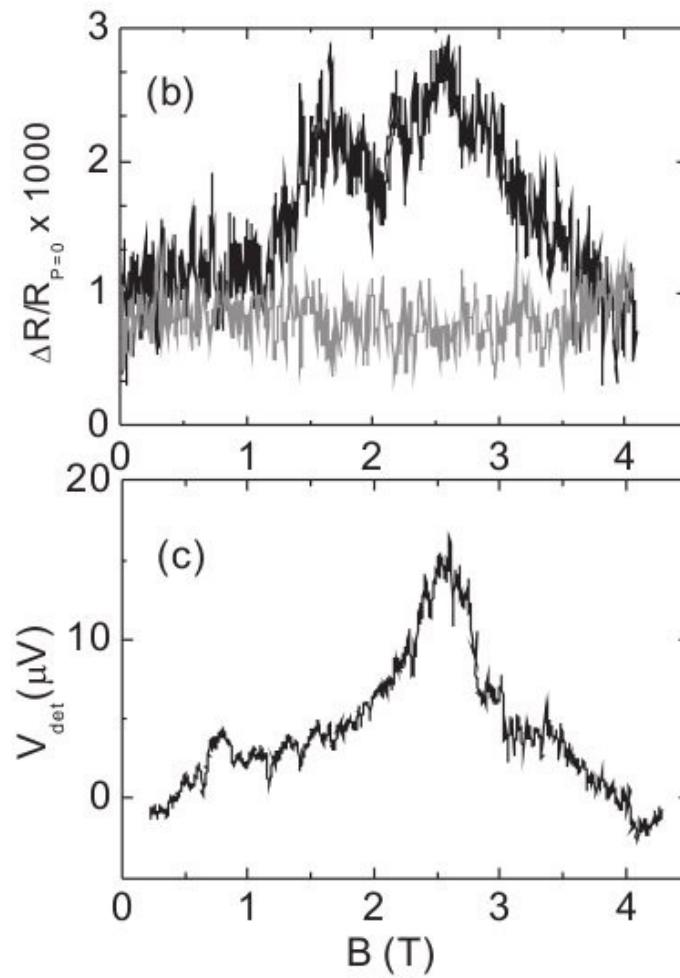
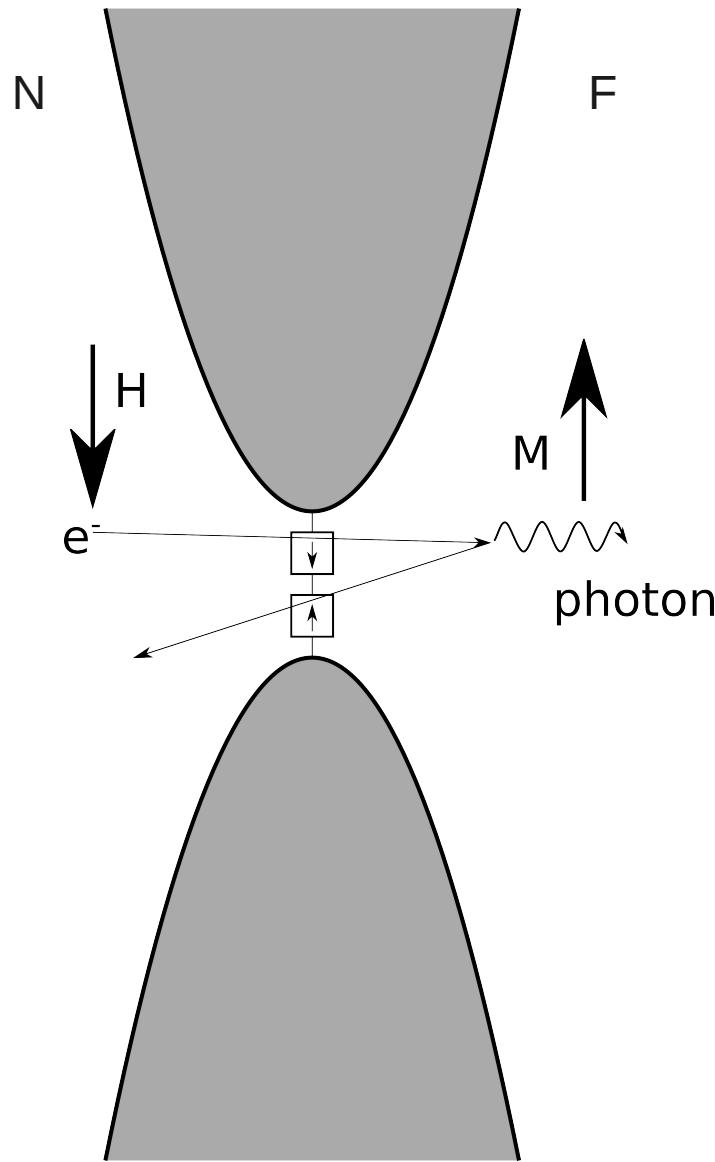
Point contact spectroscopy

Hard magnets

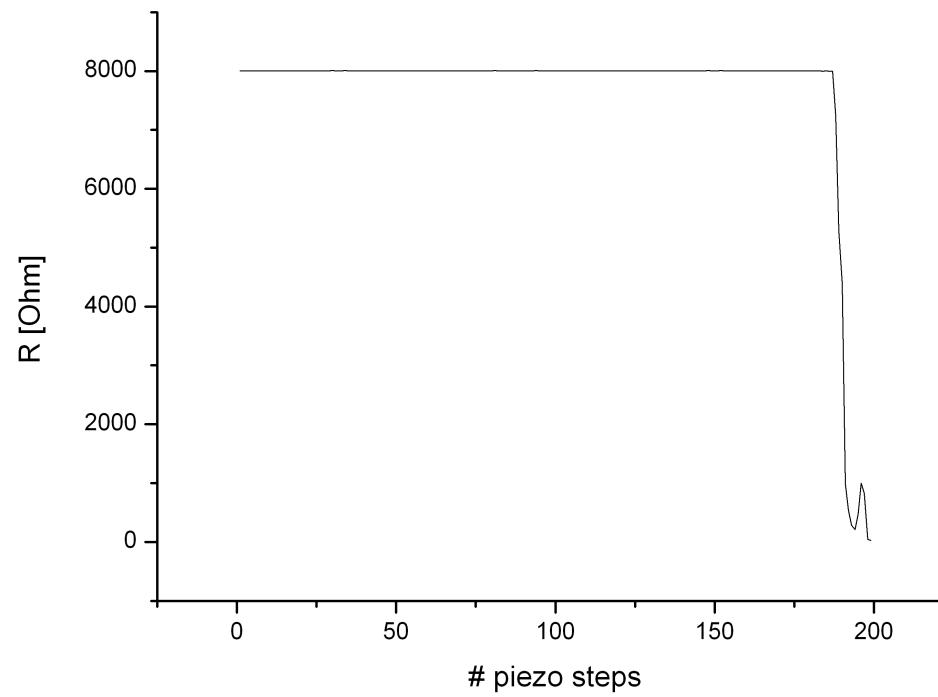
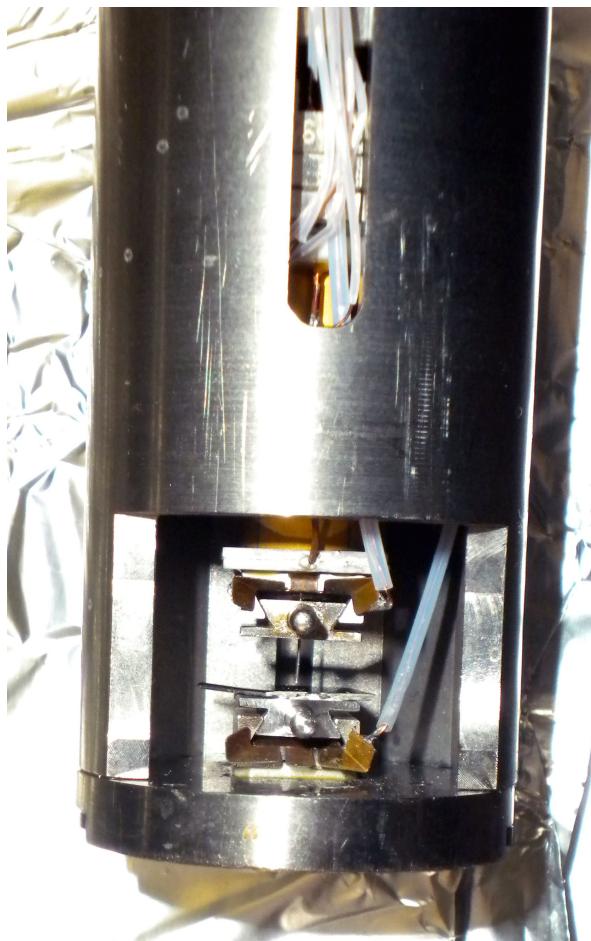
Point contact spectroscopy



FN point contact spectropy

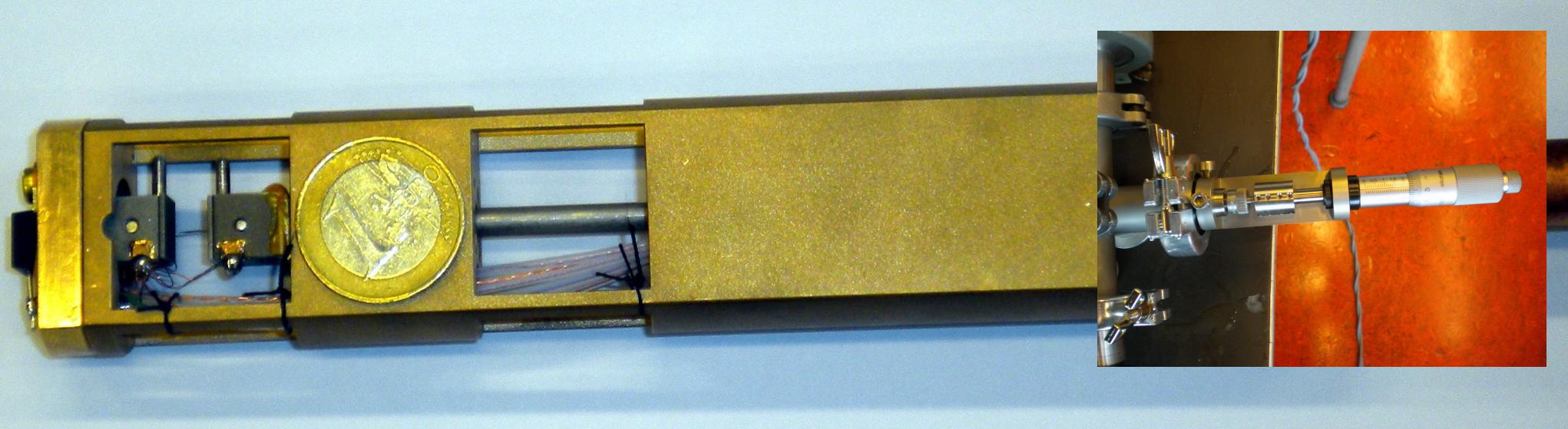


Making a point contact with attocube



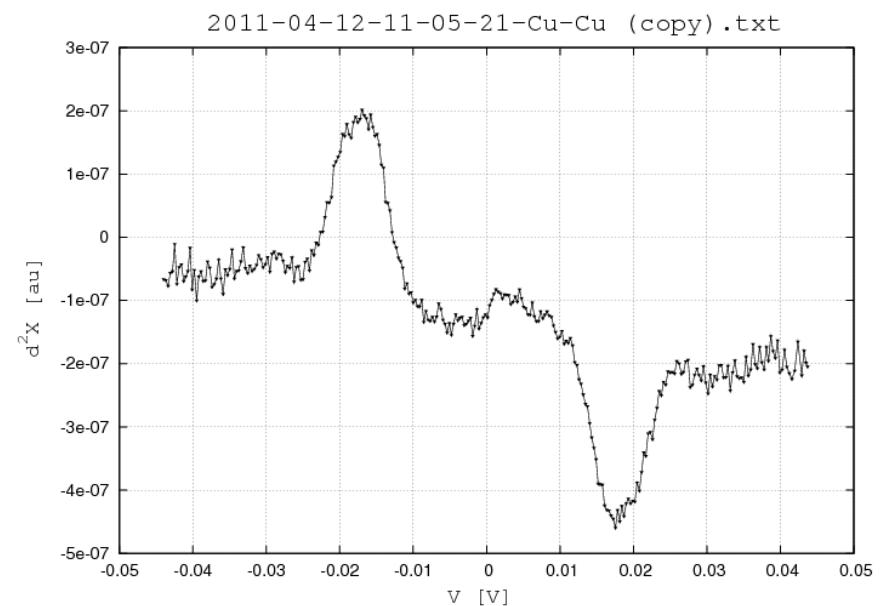
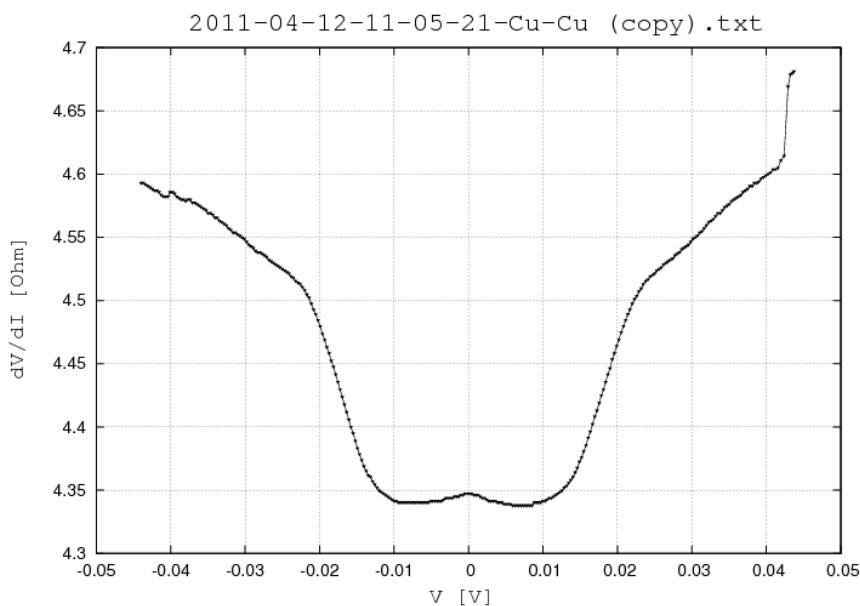
Together with A.Naylor, Leeds

Making a point contact with a micrometer screw

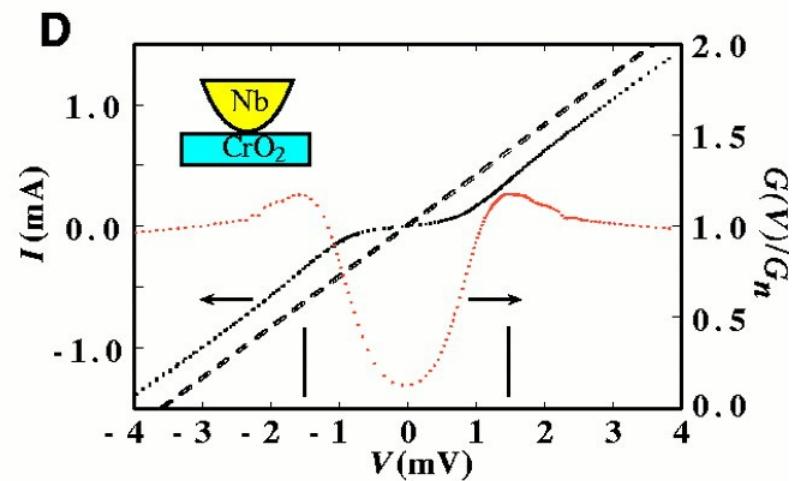
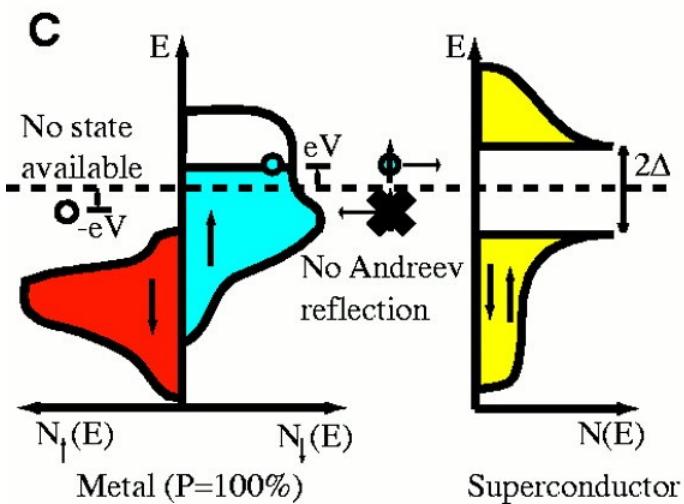
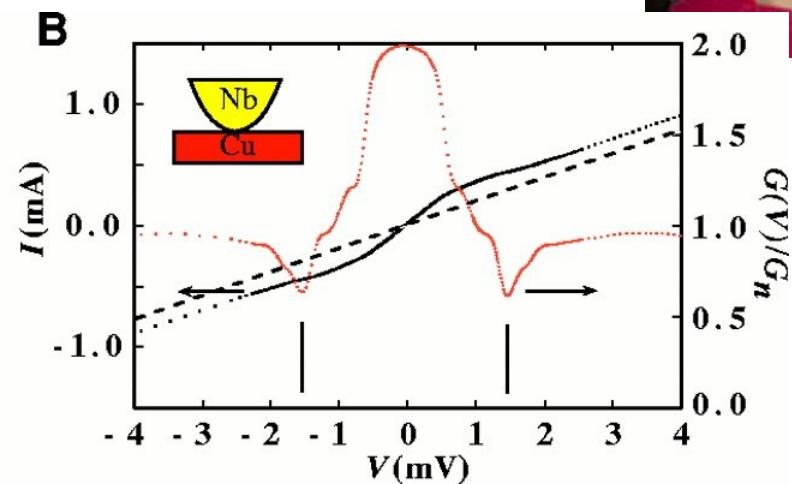
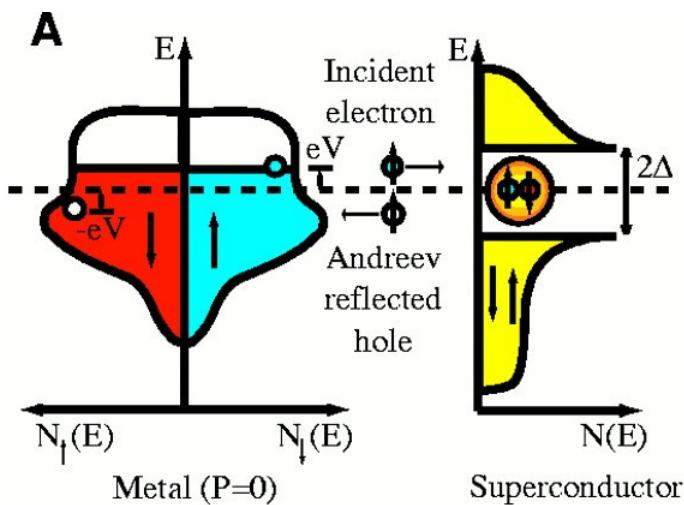


Cu/Cu point contact

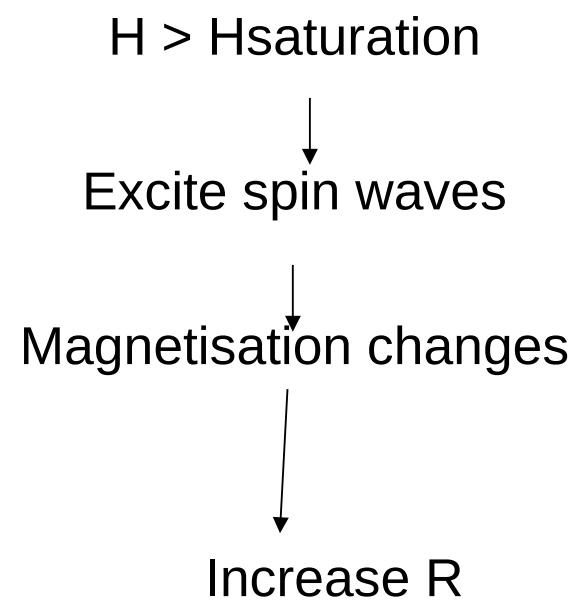
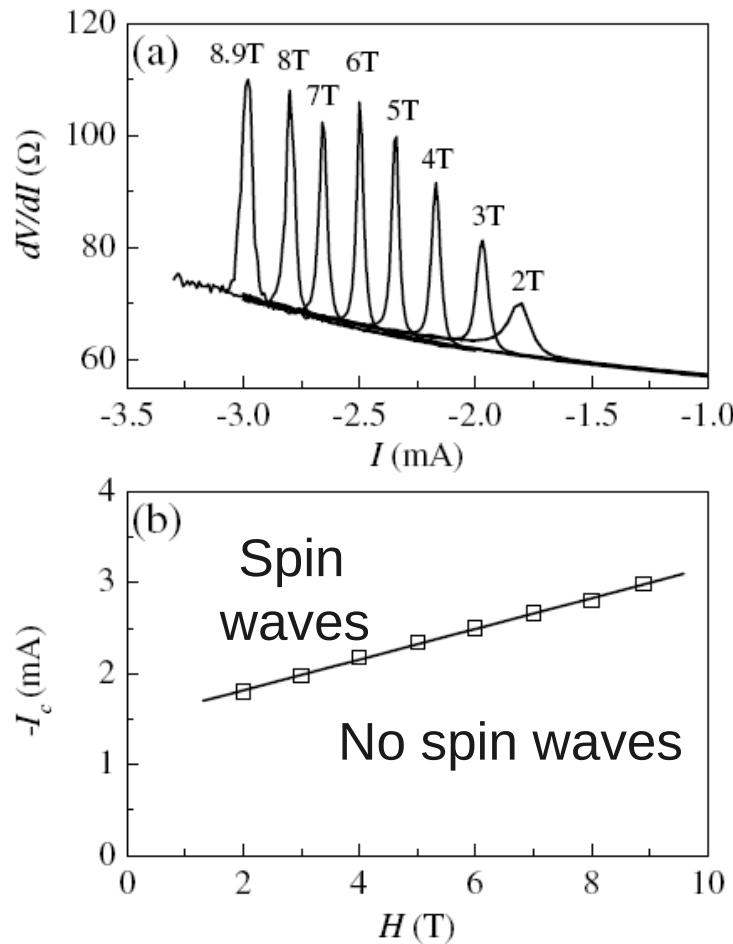
↓ Cu
Cu



Measure polarization

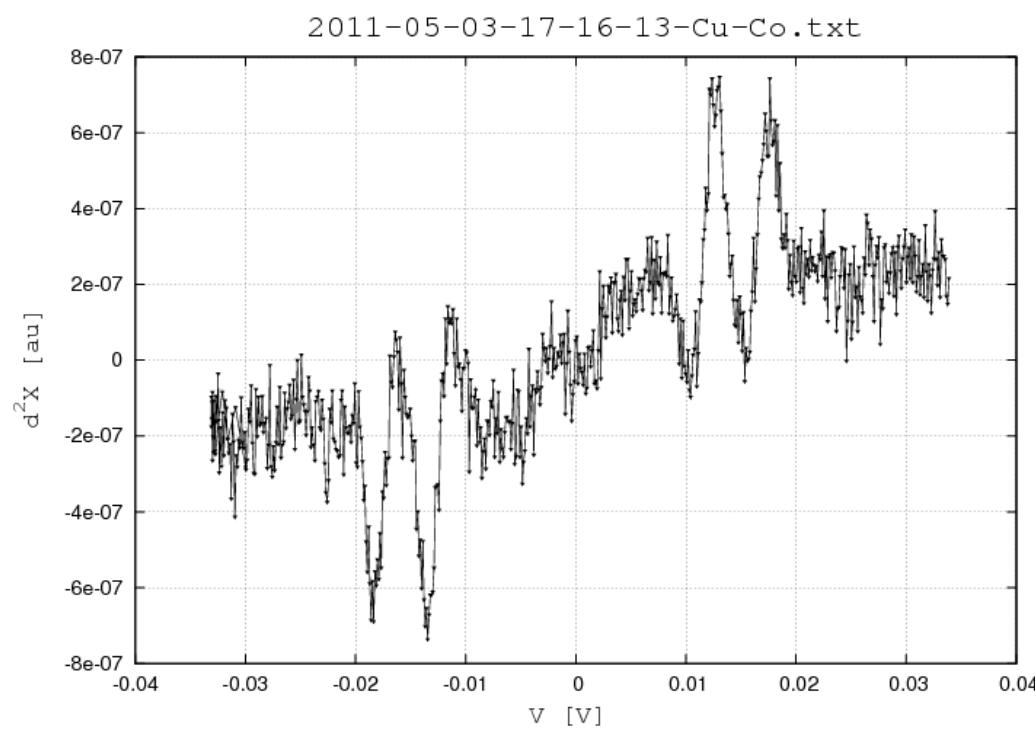
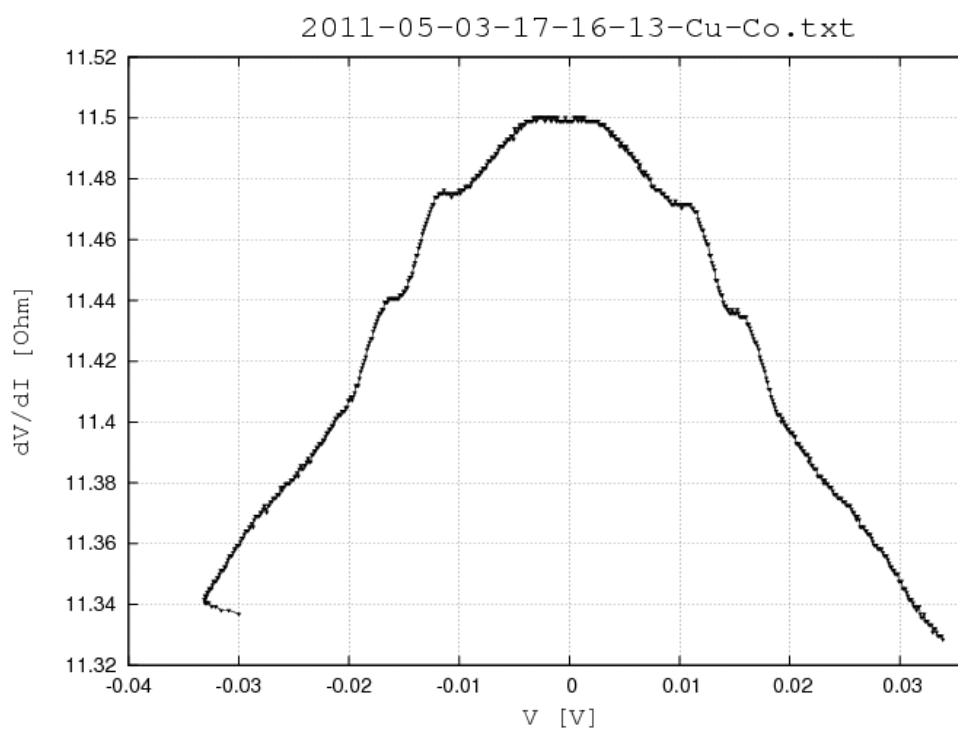


Excitations magnetic layer

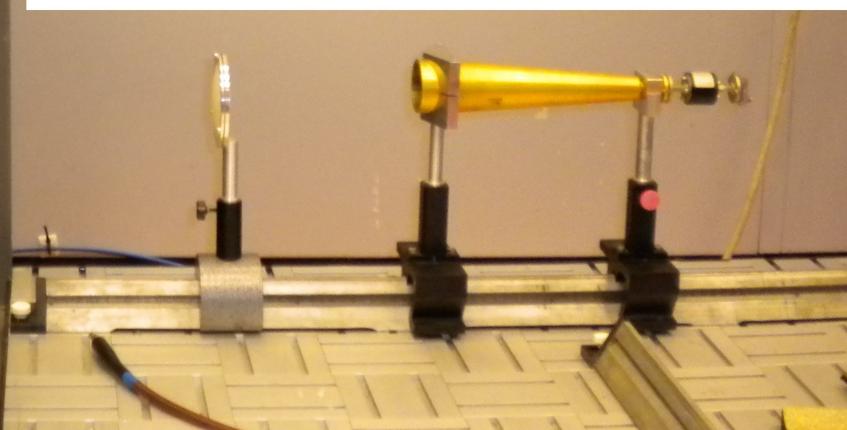


Co|Ag

PCS Cu/Co



Add radiation



Background

Point contact spectroscopy

Hard magnets

Devices, Majority F recap

If coercive field of F is (much) bigger than applied field

SmCo₅, AlNiCo, Nd₂Fe₁₄B

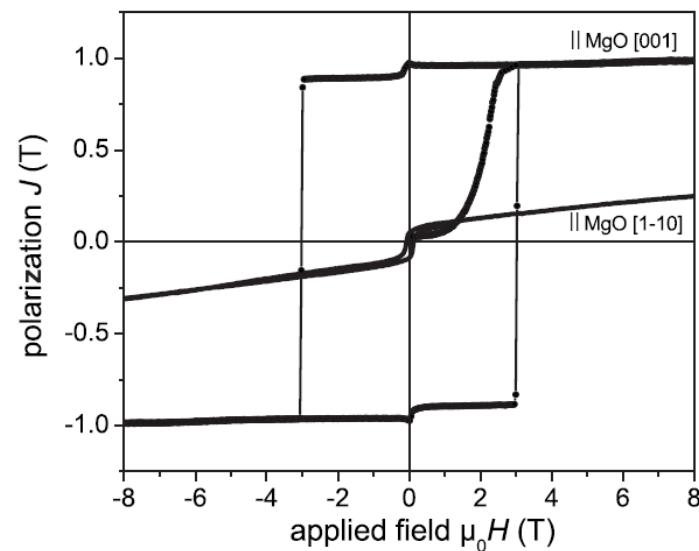
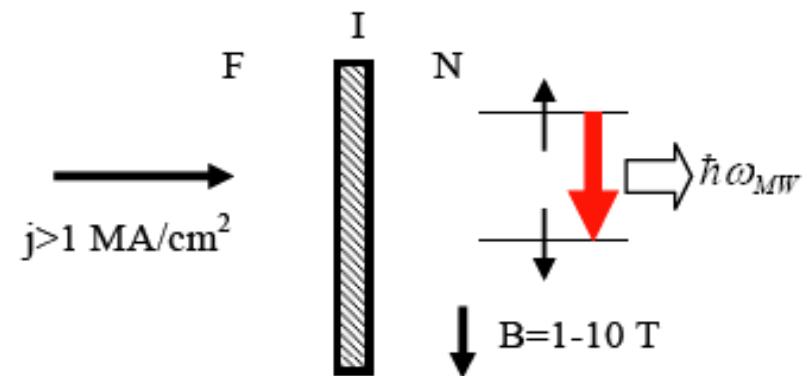


FIG. 2. Magnetic hysteresis of a SmCo₅ film measured along the easy magnetization axis (||MgO[001]) and along the in-plane hard axis (||MgO[1–10]).



Recipe to make a hard magnet

- Stoner-Wohlfarth criterium:

$$E = K \sin^2(\theta - \phi) - \mu_0 H M_s \cos \phi$$



strong uniaxial magnetic anisotropy K

-> crystallography, shape, ...



maximize B_{remnance}

-> crystal, epitaxial, ...



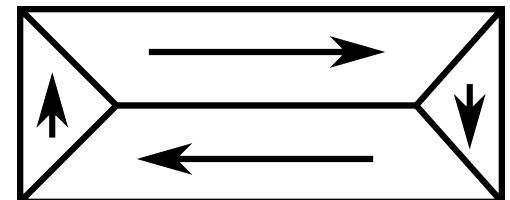
eliminate domain walls

-> single domain, pin domains, ...

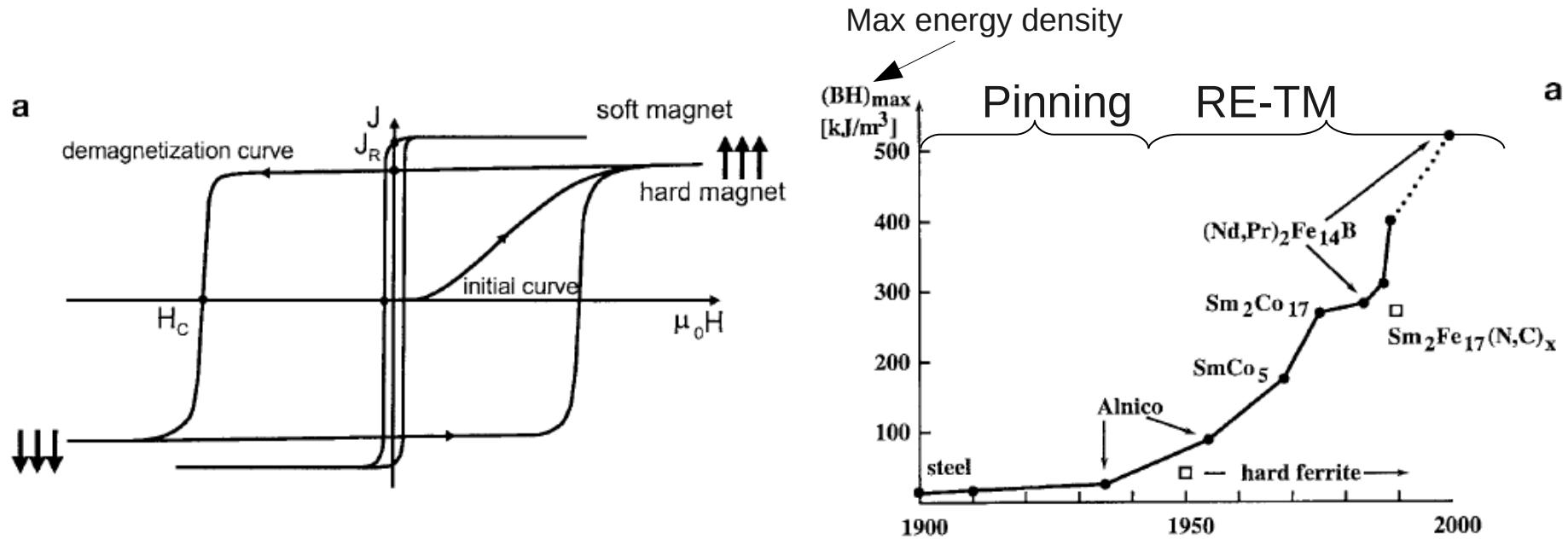


minimize exchange coupling between domains

-> nonmagnetic defects, ...



Hard magnets



Thermally activated switching:

$$\tau \propto \exp \frac{K_u V_{grain}}{k_b T}$$

10 years stable

$$\frac{K_u V_{grain}}{k_b T} > 60$$

A Family of New Cobalt-Base Permanent Magnet Materials

K. STRNAT, G. HOFFER, J. OLSON, AND W. OSTERTAG

Air Force Materials Laboratory, Dayton, Ohio

AND

J. J. BECKER

General Electric Research and Development Center, Schenectady, New York

The magnetocrystalline anisotropy of several intermetallic phases of the type RCO_5 ($R = Y, Ce, Pr, Sm, Y$ -rich and Ce -rich mischmetals) has been investigated, and it is concluded that these alloys are promising candidates for fine-particle permanent magnets. They have extremely high uniaxial anisotropy ($K = 5.4$ to 7.7×10^7 erg/cm 3), single easy axis, high saturation ($B_s = 8500$ to $11\,200$ G) and Curie point ($t_c = 464^\circ$ to $747^\circ C$). Approximate upper limits for the possible energy product lie between 18 and 31.3 MGoe. Experimentally, coercive forces of over 8000 Oe and $(BH)_{max} = 5.1$ MGoe have been observed in $SmCO_5$ merely ground at room temperature. Grinding of YCO_5 and $(Ce-MM)CO_5$ produces an increase of MH_c to 2200 and 2700 Oe, respectively, followed by a decrease as particle size continues to decrease.

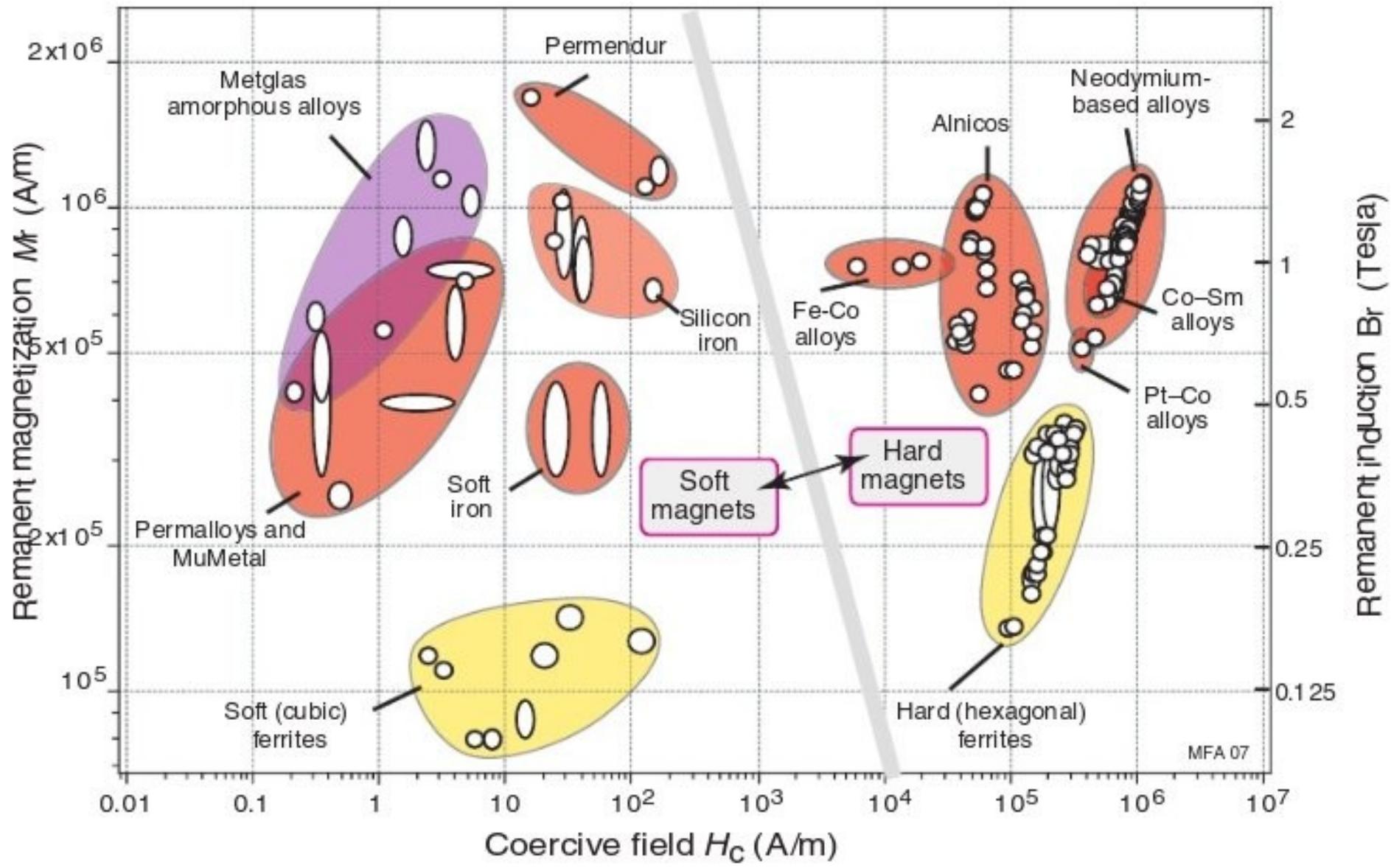
Rare earth transition metals

- Nd
- Fe
- Pr
- Co
- Sm

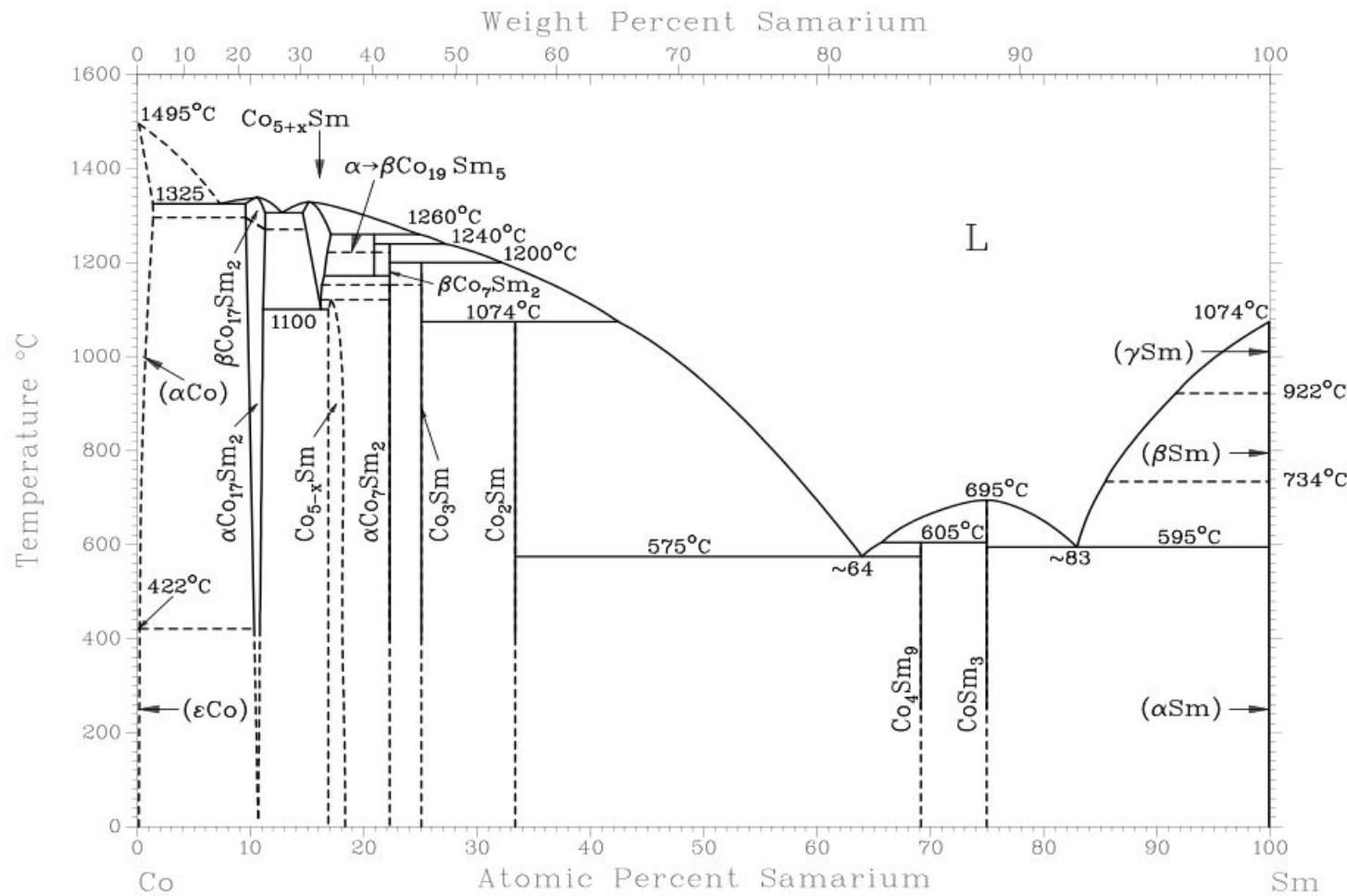
Combine

- High saturation polarization and Tcurie 3d TM
- high crystal anisotropy RE

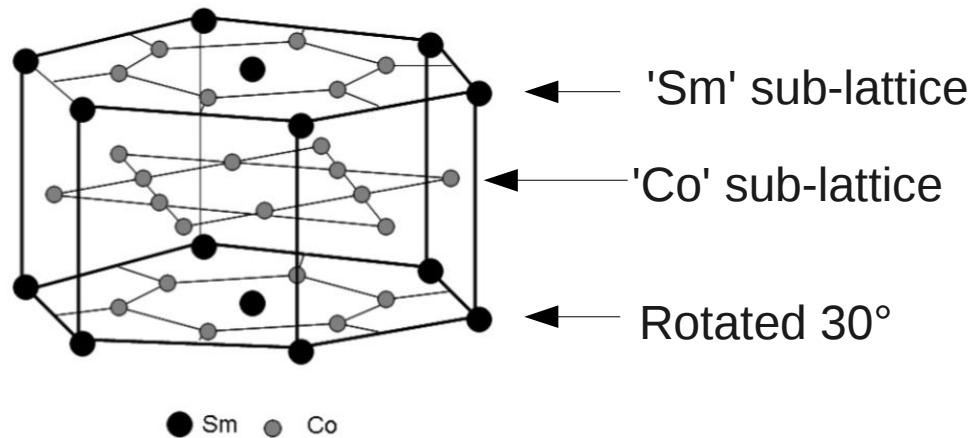
Hard magnets



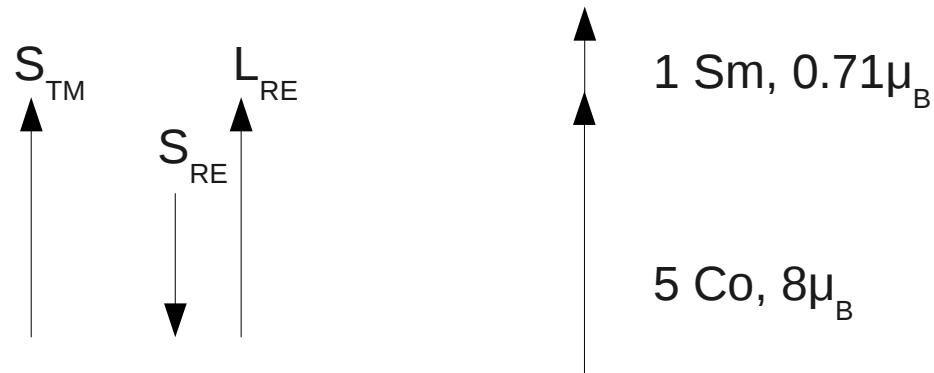
Phase diagram Sm_xCo_y



SmCo_5



How to get such a huge magnetization?



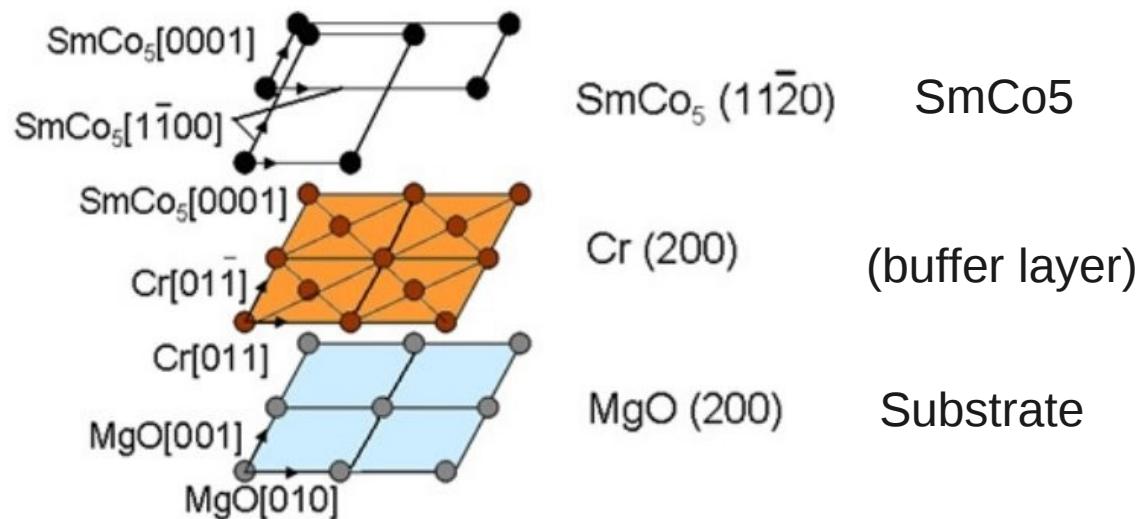
$$M_{\text{exp}} = 7.27 \mu_B$$
$$M_{\text{calc}} = 8.71 \mu_B$$

Other systems?

1																		18
1	H	2																He
2	Li	Be																Ne
3	Na	Mg	3	4	5	6	7	8	9	10	11	12	13	14	15	16	17	
4	K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	5	6	7	8	9	
5	Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	13	14	15	16	17	
6	Cs	Ba	Ln	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	31	32	33	34	35	
7	Fr	Ra	Ac										36					
				57	58	59	60	61	62	63	64	65	66	67	68	69	70	71
				La	Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb	Lu

- Anisotropy Co biggest
- Only easy axes of Sm, Er, Tm combines well with Co
- If L & S parallel, total moment antiparallel -> ferrimagnetism (GdNi)

SmCo_5 thin film growth

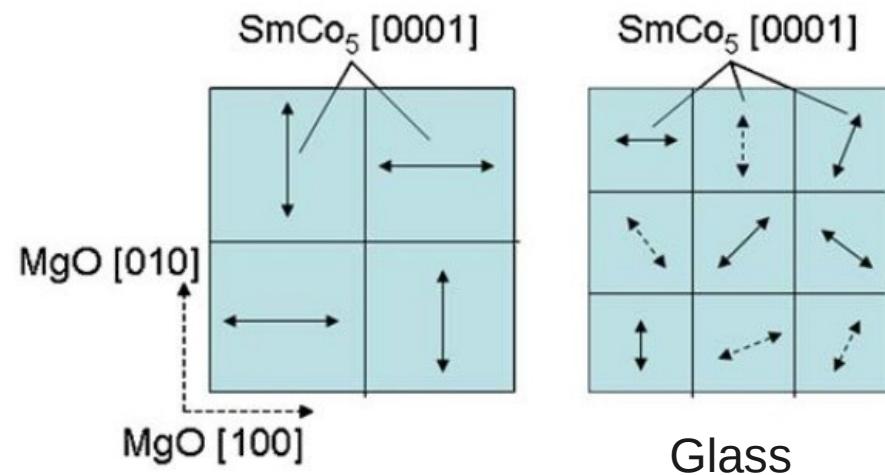


Substrate

Single crystal to get desired texture

- MgO (100), MgO(110)
- Si (110)

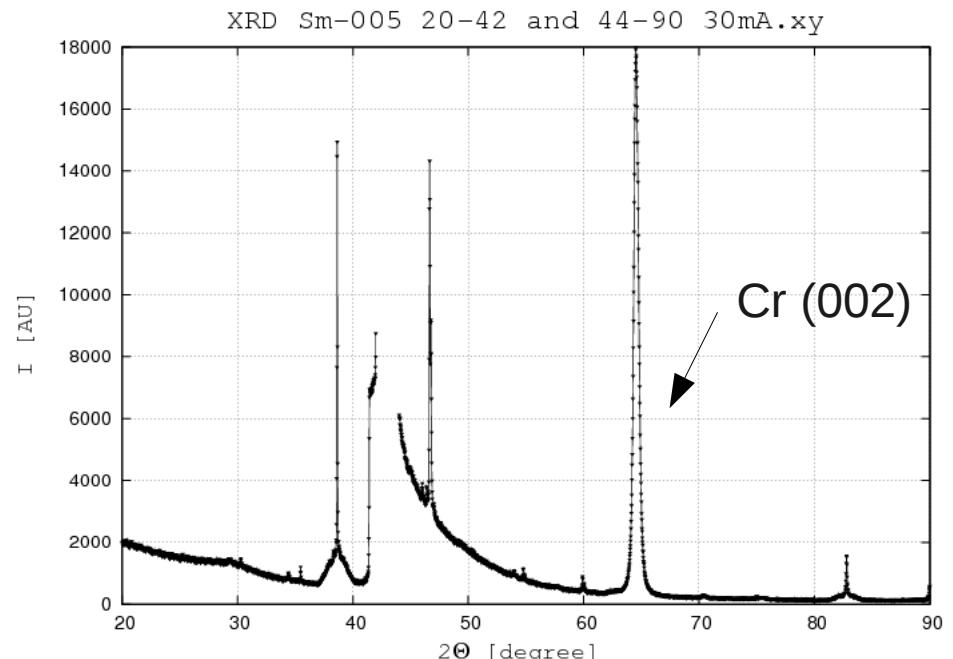
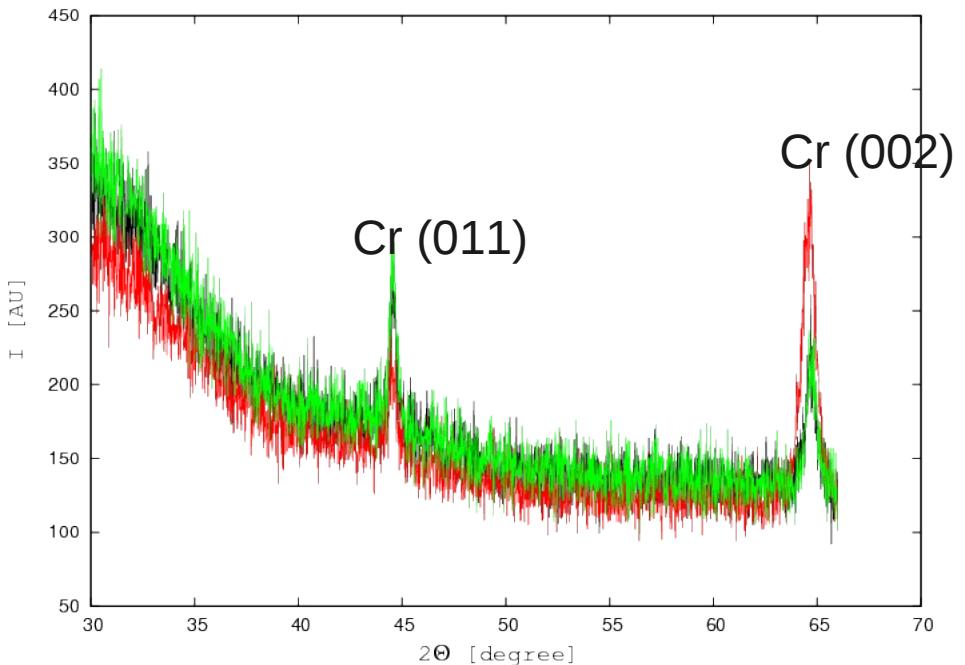
Amorphous glass



Buffer layer

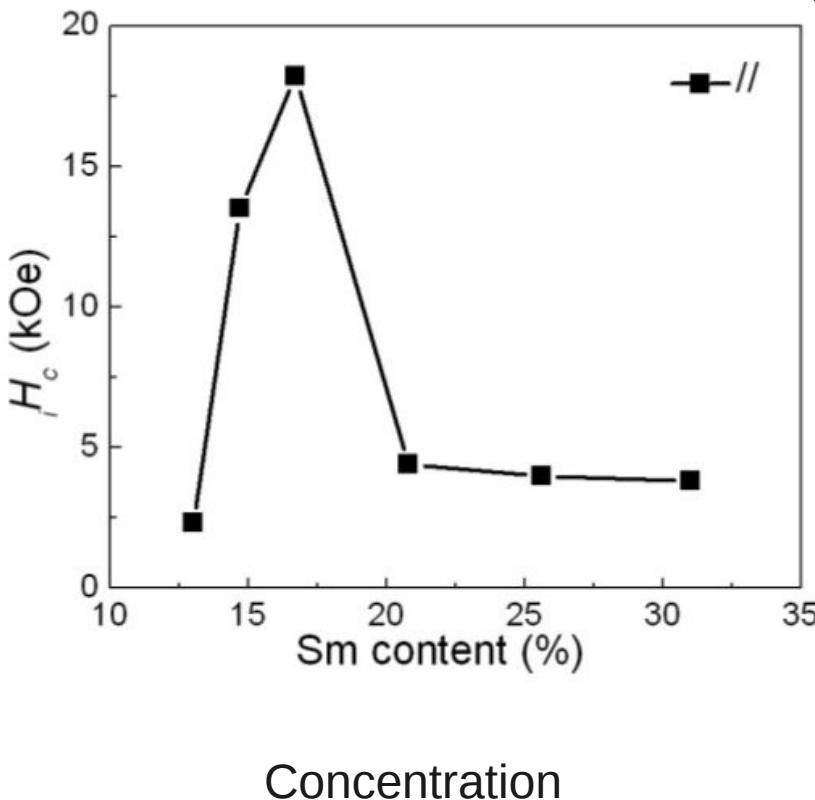
Use Cr buffer layer:

- Lattice mismatch SmCo₅ | MgO 7 %
lattice mismatch Cr | MgO ~4 %
- Decrease elastic distortion
- Cr produces:
 - dense film
 - small grains
 - smooth surface

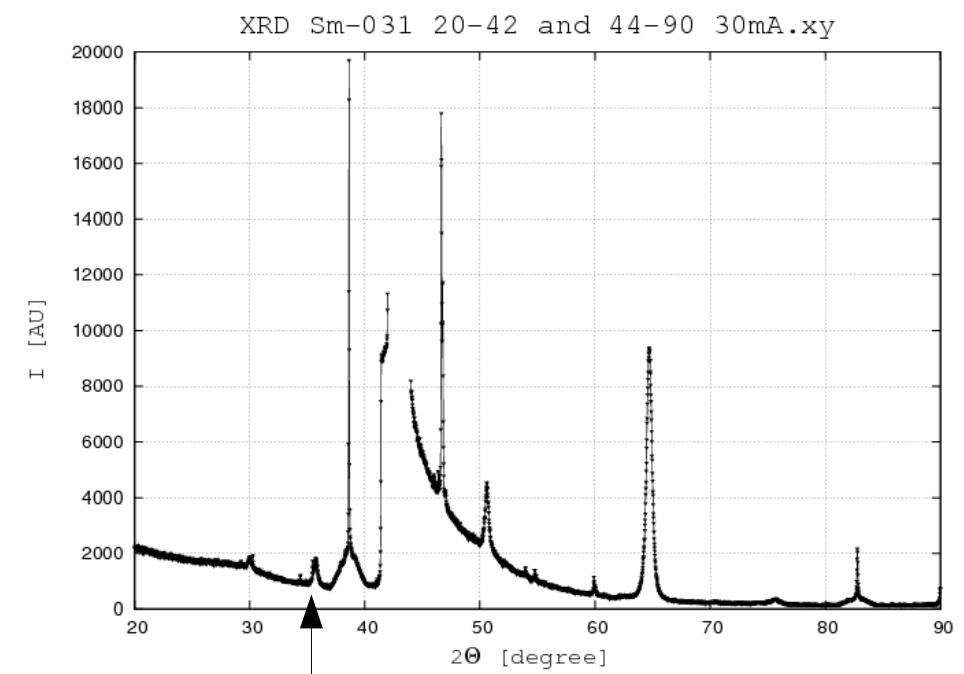


SmCo_5 film

- DC sputtering composite target $\text{Sm}_{20}\text{Co}_{80}$
- [Sputter $\text{Sm}(\text{Co,Cu})_5$]*



Concentration

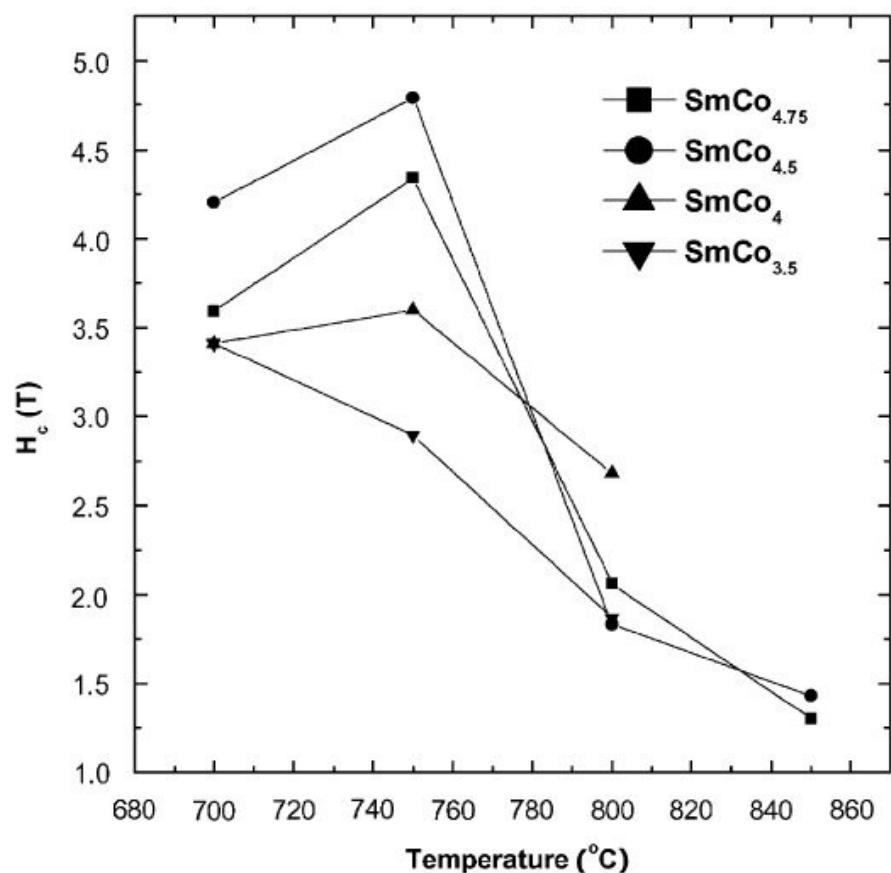


SmCo5 (11-20)

Crystal structure

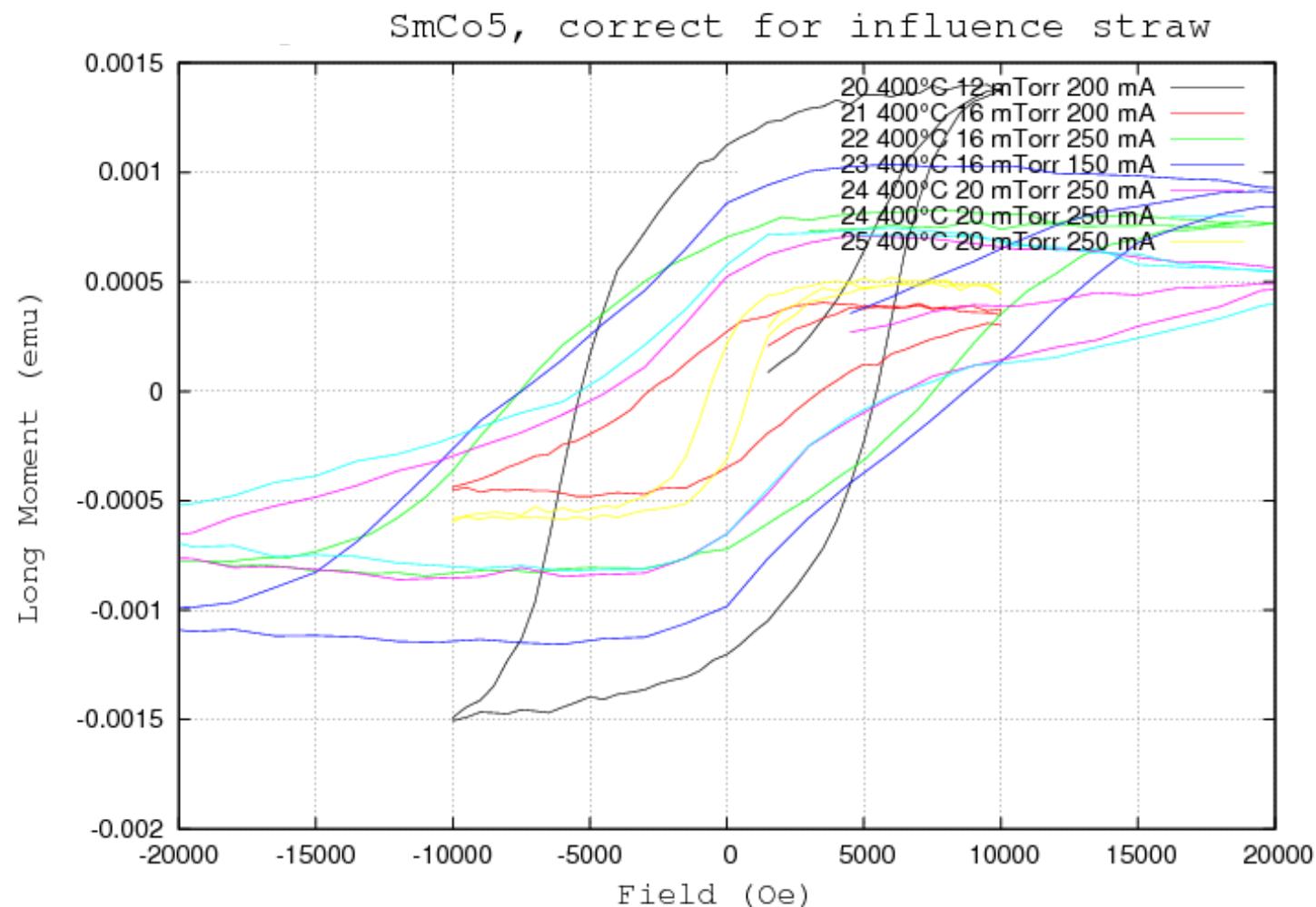
* J.Zhang et al, Jmmm 310, 1

Annealing

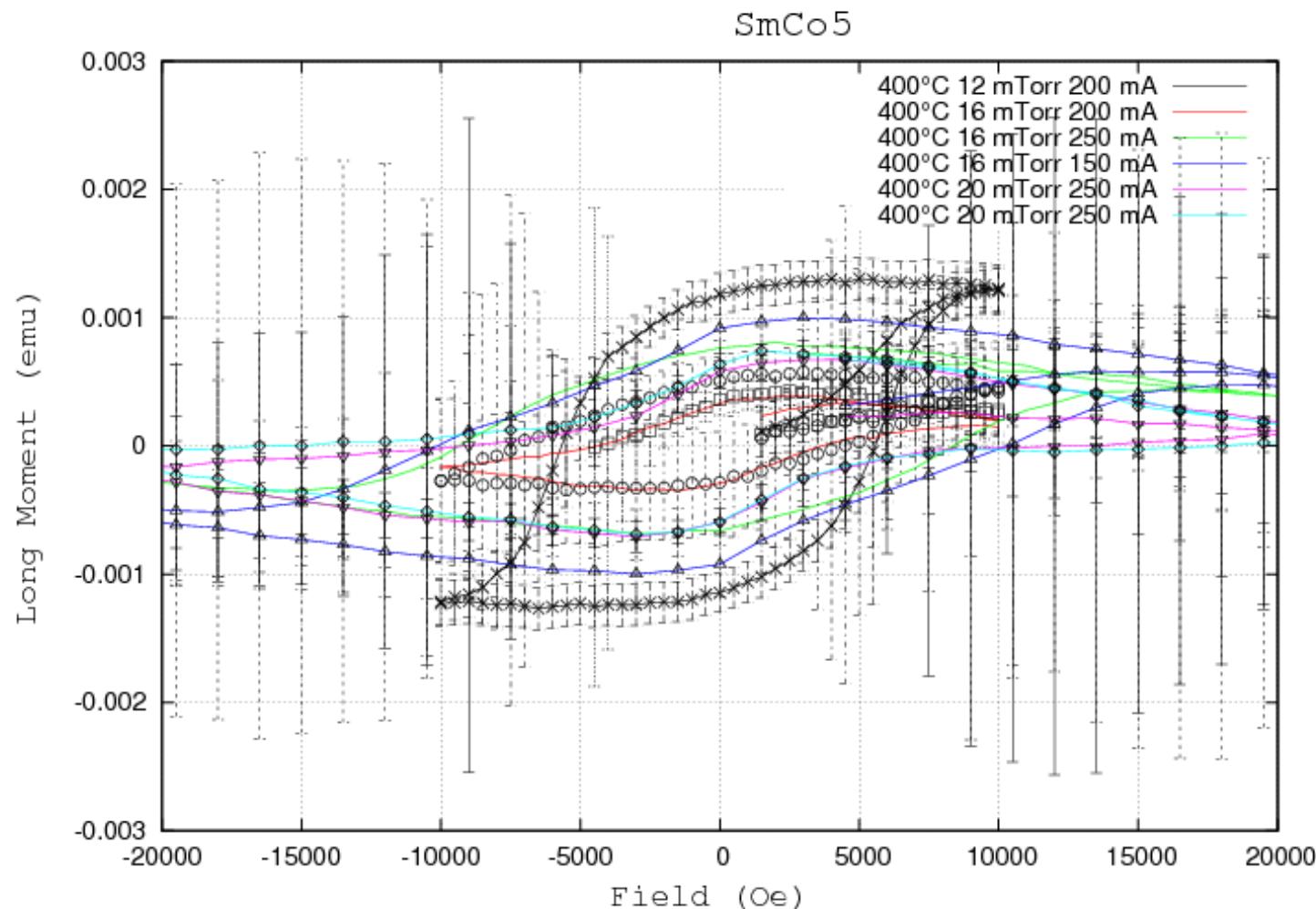


- Rapid thermal annealing
- Diffusion between layers
- Change crystallography

Coercive fields grown SmCo₅



Coercive fields, with errorbar, grown SmCo₅



Outlook

- Grow SmCo_5 & measure with squid
- PCS FN structures and observe STT (Stefano)
- Apply radiation
- Measure V_{ish} during spin pumpings
- Spin pump GdNi (Hiske)

Spin pumping

